

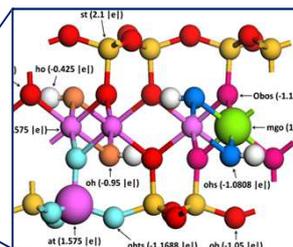
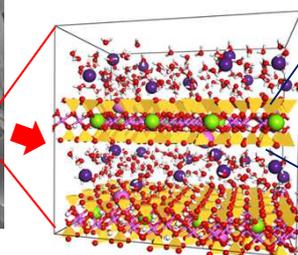
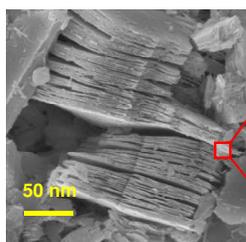
**NE-M1-PRI12ENP – Integrated Nuclear Engineering Project**

# **Molecular Modeling of Materials for Nuclear Waste Disposal Applications**

## **Lecture 4 - Inorganic Interfaces**

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<https://www.imt-atlantique.fr/en/person/andrey-kalinichev>

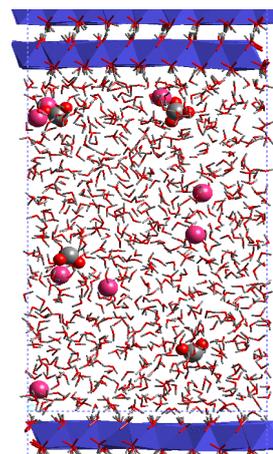


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"Molecular modeling of materials for nuclear waste disposal applications"

1

## **Interfaces of Aqueous Solutions with Inorganic Substrates**

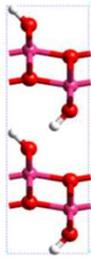
- ◆ Molecular models of aqueous interfaces with inorganic materials
- ◆ Rigid vs flexible substrate
- ◆ Structure, energetics, and dynamics of aqueous interfaces
- ◆ Hydrophobic and hydrophilic surfaces
- ◆ Local electrostatic fields and orientation of surface H<sub>2</sub>O molecules



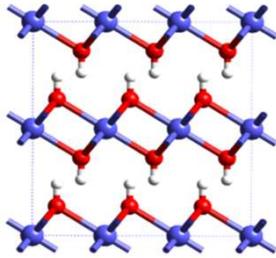
NE-M1-PRI12ENP – Integrated Nuclear Engineering Project, February-June 2026  
"Molecular modeling of materials for nuclear waste disposal applications"

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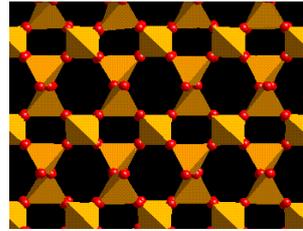
## Atomistic Models of Simple (Hydr)Oxides



Boehmite  $\gamma\text{-Al(OOH)}$

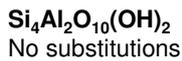


Portlandite  $\text{Ca(OH)}_2$



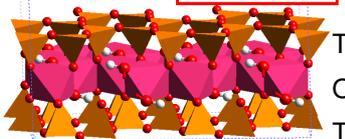
Quartz  $\text{SiO}_2$

## Structure and Composition of Complex Clays



Non-swelling

Pyrophyllite



Illite (muscovite)

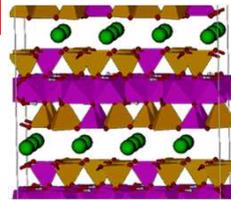


Many substitutions

Many interlayer ions

1.0e charge per  $\text{O}_{10}(\text{OH})_2$

Non-swelling



Smectite (montmorillonite)

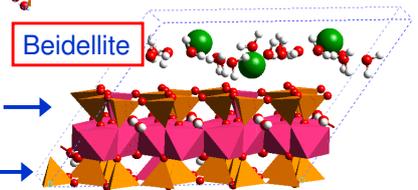
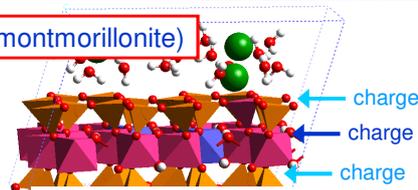
Diocahedral Smectites



Some tetrahedral Al sub  
Mostly octahedral Mg sub

Swelling

0.375e charge per  $\text{O}_{10}(\text{OH})_2$



Beidellite



Mostly tetrahedral Al sub

Swelling

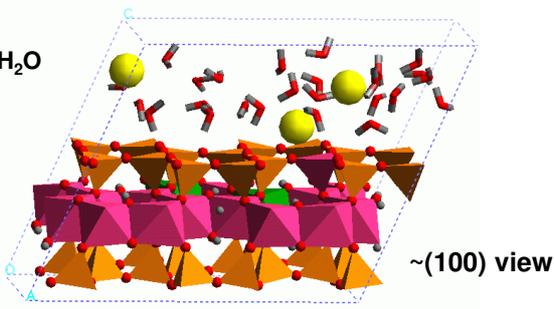
charge →

charge →

## Simulation of Complex Clays

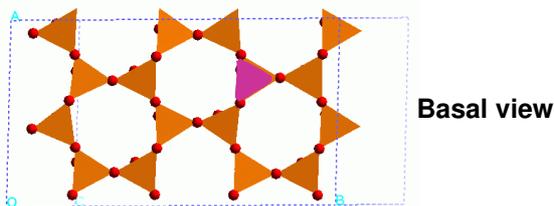
Wyoming Montmorillonite  
 $\text{Na}_3(\text{Si}_{31}\text{Al})(\text{Al}_{14}\text{Mg}_2)\text{O}_{80}(\text{OH})_{16} \cdot n\text{H}_2\text{O}$

- No refined structures
- Cryptocrystalline
- Stacking disorder
- Complex compositions
- Site disorder



MD with *NPT*-ensemble  
 50 psec

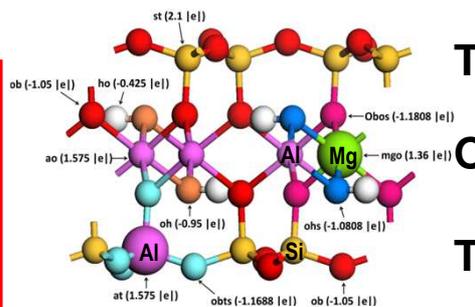
Six-member rings  
 of  $\text{SiO}_4$  tetrahedra



## ClayFF Construction and Parametrization

1. No explicit bonds – quasi-ionic
2. LJ parameters for all oxygen atoms are assumed to be equal to  $\text{Ow}$  for SPC water

Both assumptions are great simplifications of reality, but they seem to work quite well

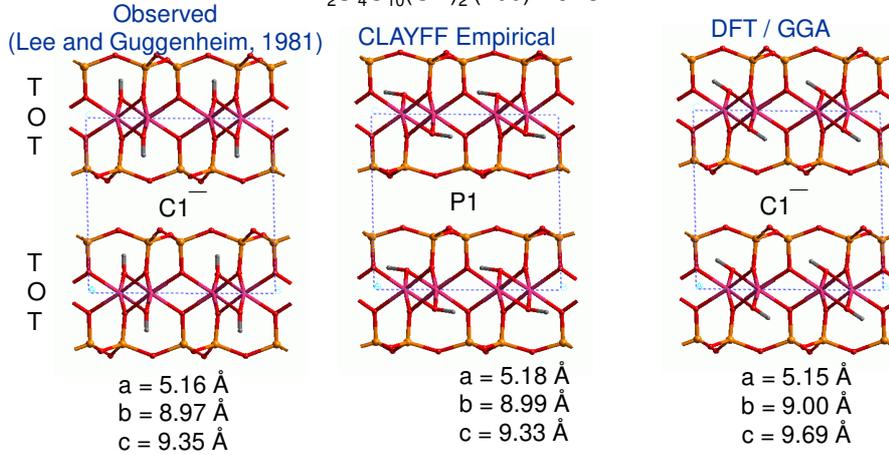


- Accurate determinations of partial charges are required to represent charge distributions of interlayer and external surfaces where electrostatic forces control sorption and transport processes
- Atomic charges derived from DFT calculations for cluster and periodic models of **simple oxide and hydroxide phases** (brucite, gibbsite, kaolinite, quartz...)
- Allows for **charge delocalization** among coordinating oxygens for substitutions

**ClayFF:** Cygan, Liang, Kalinichev, *J. Phys. Chem. B*, **108** 1255-1266 (2004)  
 Cygan, Greathouse, Kalinichev, *J. Phys. Chem. C*, **125** 17573-17589 (2021)

## Modeling of Pyrophyllite

$\text{Al}_2\text{Si}_4\text{O}_{10}(\text{OH})_2$  (100) views

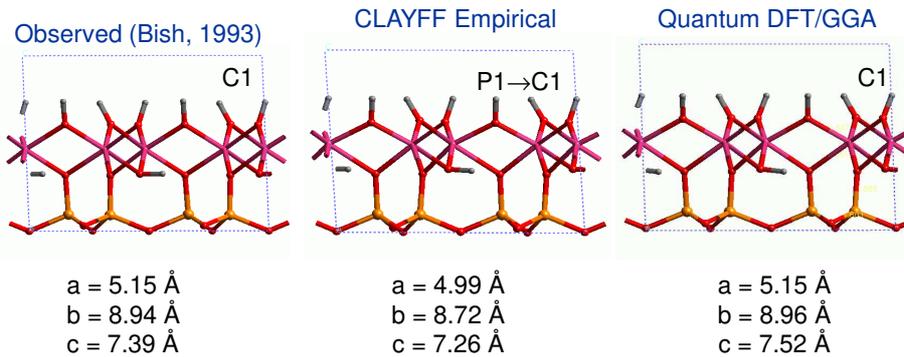


<b>Unit cell parameters:</b>	$\Delta < 0.5\%$	$\Delta < 3\%$
<b>Bond lengths:</b>	$\Delta < 4\%$	$\Delta < 2\%$

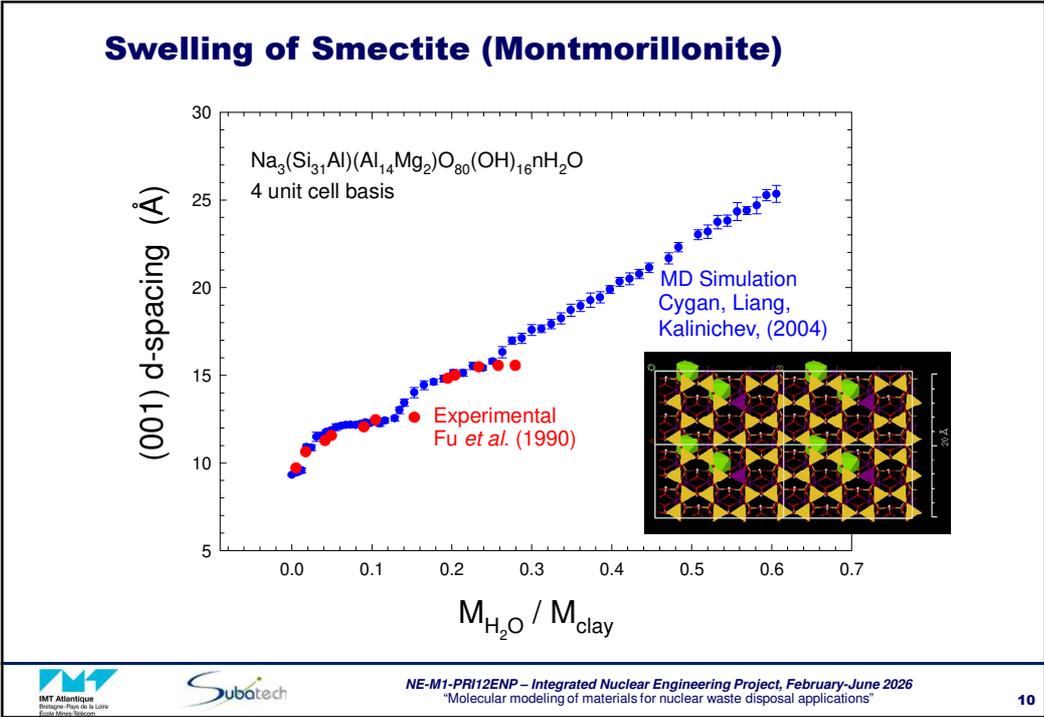
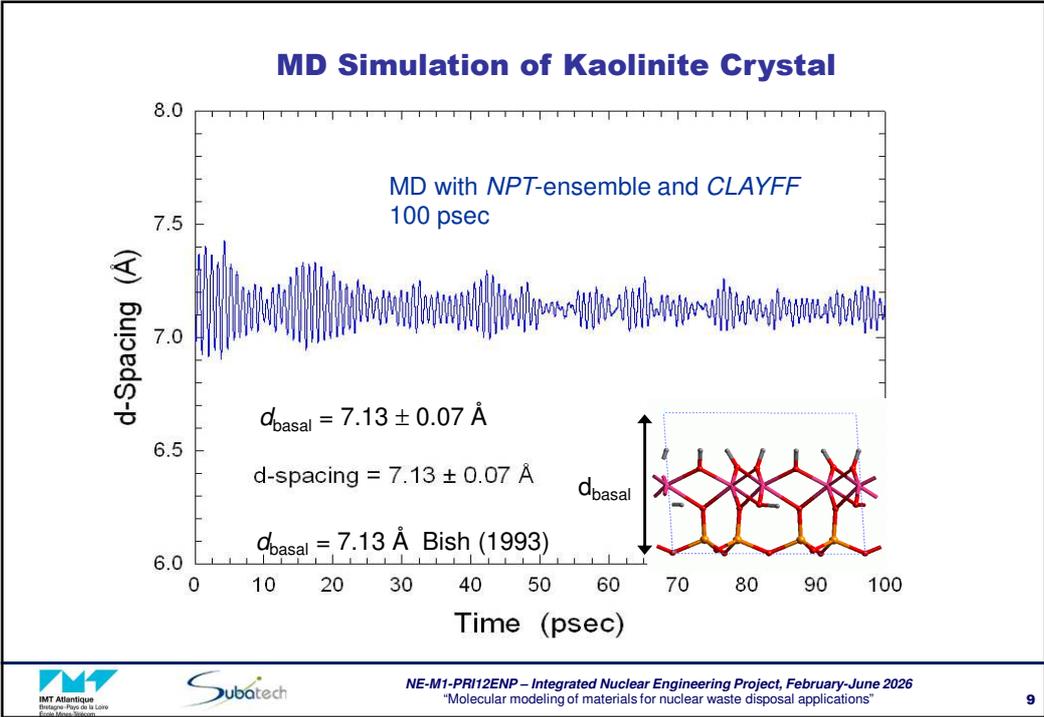
## Modeling of a Simple Clay Crystal: Kaolinite

$\text{Al}_2\text{Si}_2\text{O}_5(\text{OH})_4$  (100) views

(Cygan et al., 2004)

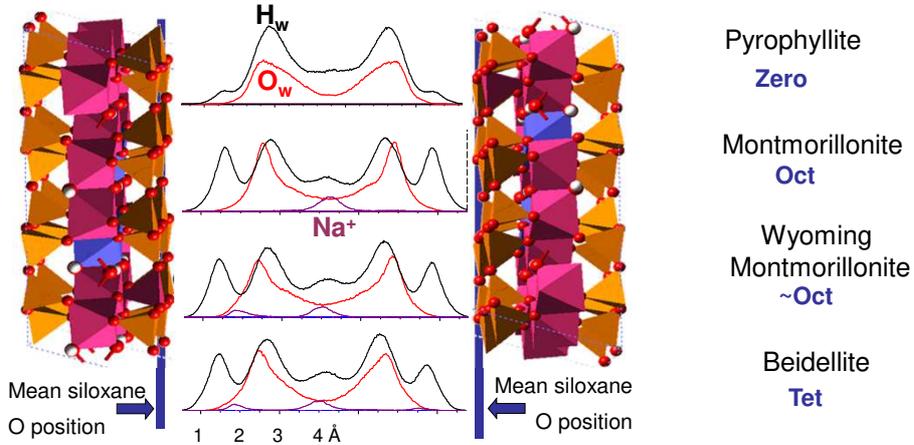


<b>Unit cell parameters:</b>	$\Delta < 3\%$	$\Delta < 2\%$
<b>Bond lengths:</b>	$\Delta < 4\%$	$\Delta < 3\%$



## Association of Metal Cations with Clay Surfaces

### Two-Layer Clay Hydrates

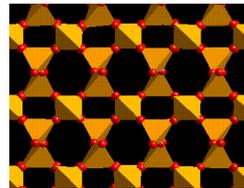
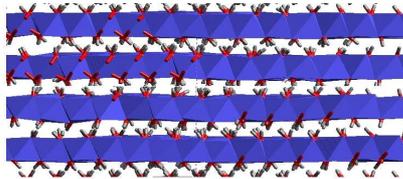


**Location of substitution sites in the clay structure can strongly influence the distribution and mobility of adsorbed ions and H<sub>2</sub>O molecules**

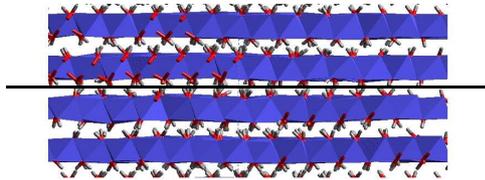
## Modeling Substrate-Water Interfaces (1)

### Building a mineral-solution interface model

- Build a model of bulk crystal

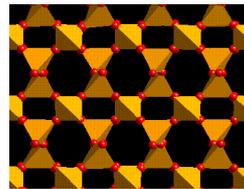


## Modeling Substrate-Water Interfaces (2)

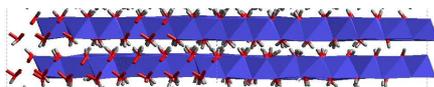


### Building a mineral-solution interface model

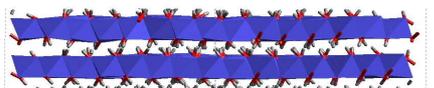
- ◆ Build a model of bulk crystal
- ◆ Define the cleavage plane



## Modeling Substrate-Water Interfaces (3)



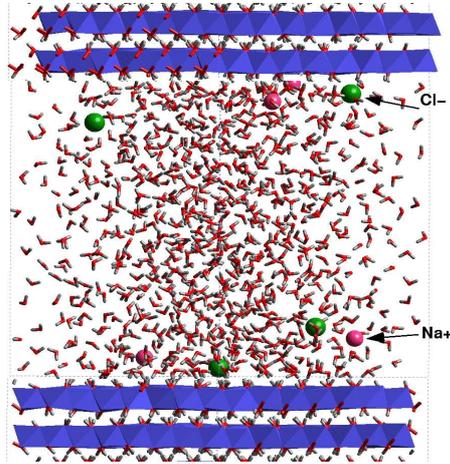
Most hydrated oxide surfaces are at least partially protonated (terminated by OH-groups) under normal pH



### Building a mineral-solution interface model

- ◆ Build a model of bulk crystal
- ◆ Define the cleavage plane
- ◆ Cleave the crystal to create a slit-like nano-pore

## Modeling Substrate-Water Interfaces (4)

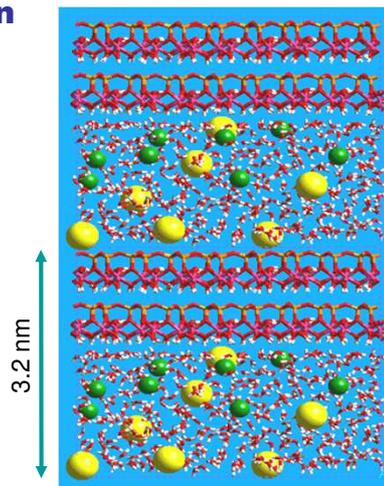


### Building a mineral-solution interface model

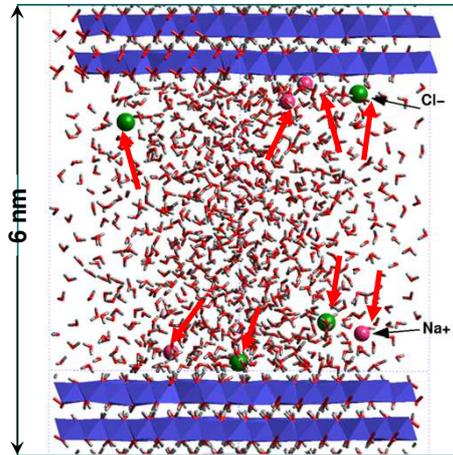
- ◆ Build a model of bulk crystal
- ◆ Define the cleavage plane
- ◆ Cleave the crystal to create a slit-like nano-pore
- ◆ Add pore solution from a preliminary bulk-solution simulation

## 1 m CsCl Aqueous Solution in Slit-Like Nanopores of Kaolinite

Layered model of interfaces in periodic boundary conditions



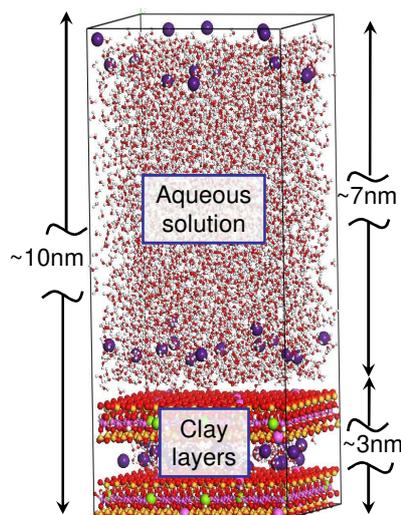
## Ionic Sorption at the Interfaces



### 3 types of surface species:

- ◆ **Inner-sphere** surface complexes (i.e., located directly on the solid-phase surface)
- ◆ **Outer-sphere** surface complexes (i.e., separated from the solid surface by one molecular layer of water)
- ◆ Ions in the **diffuse layer** or in the **bulk** solution (which still can feel the presence of the surface, but are separated from it by more than one molecular layer of water)

## MD Modeling of Clay-Solution Interfaces



### **Classical Newtonian dynamics**

- $N_{\text{tot}} \sim 3,000 - 10,000$  atoms
- $N_{\text{H}_2\text{O}} \sim 0 - 1,000$  molecules
- ClayFF force field (Cygan et al., 2004)
- $a \times b \times c \sim 3 \times 3 \times 10 \text{ nm}^3$
- Periodic boundary conditions
- NVT- or NPT-ensemble  $T=300\text{K}$ ;  $P=1 \text{ bar}$
- $t \sim 200 - 1,000 \text{ ps}$
- $\Delta t = 0.5-1.0 \text{ fs}$

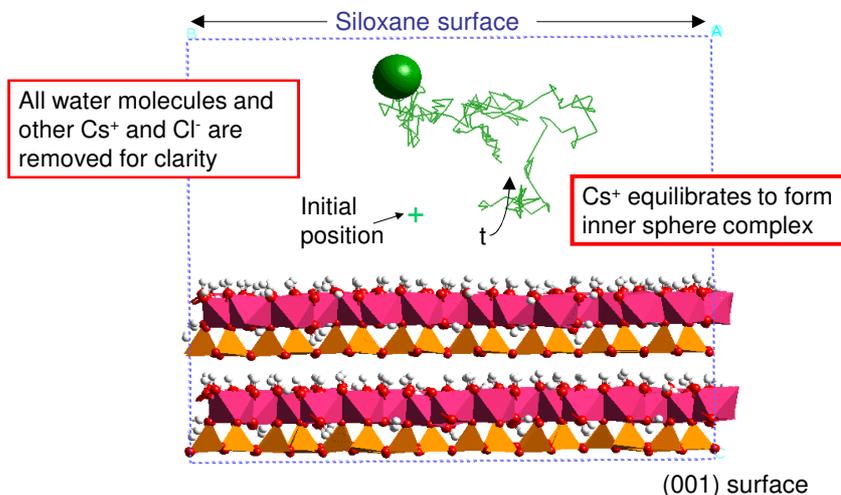
### **Solution structure:**

- ✓ Atomic density profiles ( $\perp$ )
- ✓ Atomic density surface distributions ( $\parallel$ )
- ✓ Topology of the interfacial H-bond network

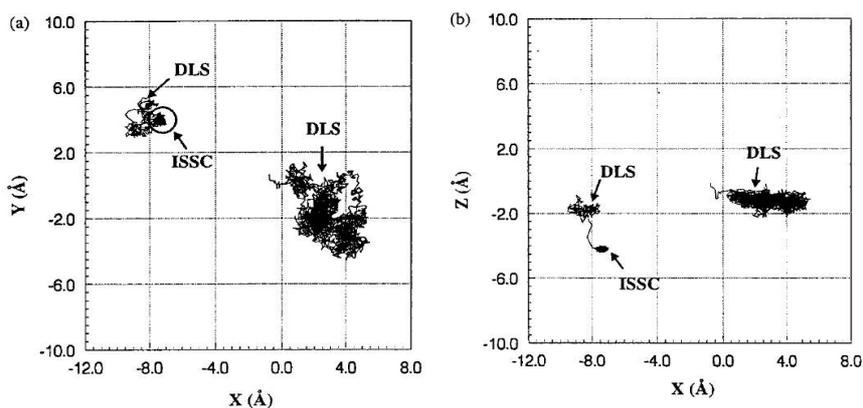
### **Dynamics:**

- ✓ Diffusion coefficients (longer time scale)
- ✓ Spectra of vibrational and rotational dynamics (shorter time scale)

## Trajectory Path of Cesium Ion at the Interface of CsCl Aqueous Solution with Kaolinite

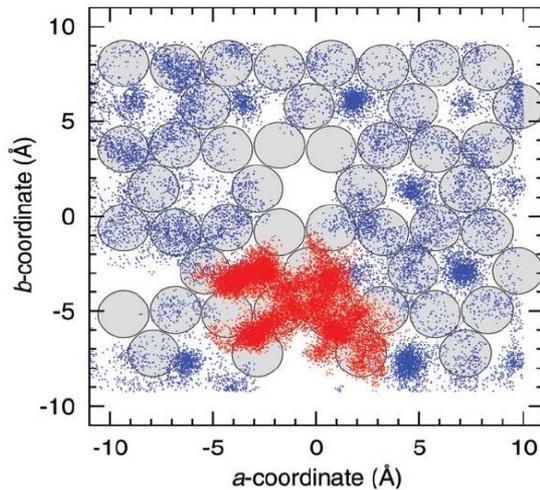


## Analysis of Structure and Energetics of Aqueous Interfaces



MD trajectory plots for Li<sup>+</sup> counterions in the three layer montmorillonite hydrate (a) XY plane; (b) XZ plane. ISSC - inner-sphere surface complex; DLS - diffuse-layer species.  
Chang et al., *Langmuir*, **13**, 2074-2082 (1997)

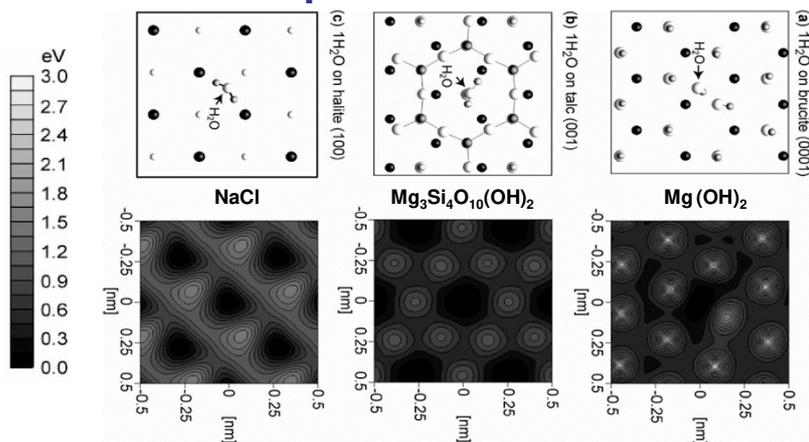
## Analysis of Structure and Energetics of Aqueous Interfaces



Center-of-mass trajectory of a  $Zn^{2+}$ -“bound”  $H_2O$  molecule (red) and a “free”  $H_2O$  molecule (blue) in the  $ab$  crystallographic plane in the 2.4 ns MD run of a hydrated Zn-vermiculite. The large grey circles denote oxygen atoms of the clay surface.

Arab et al., *PCCP*, 6, 2446-2453 (2004)

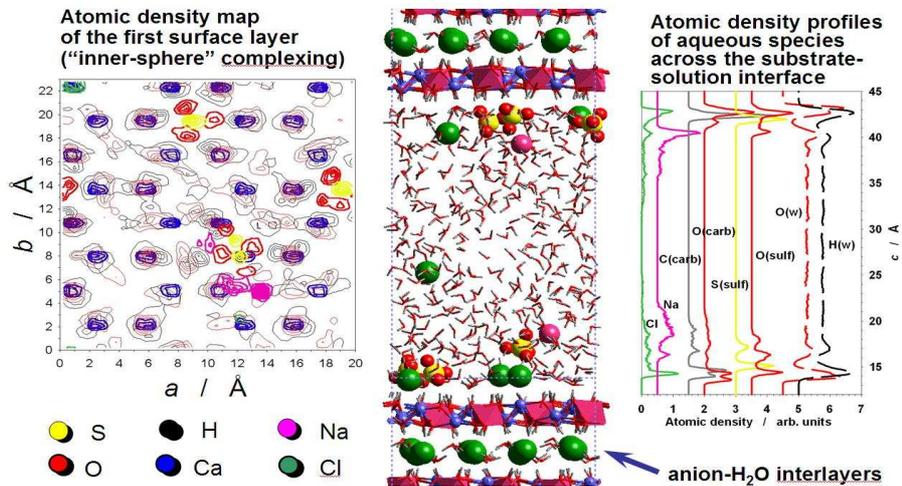
## Analysis of Structure and Energetics of Aqueous Interfaces



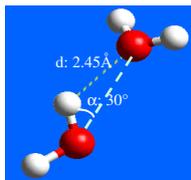
Most stable configuration (top) and contour maps of potential energy surface (bottom) for a water molecule on (a) brucite (0001), (b) talc (001), and (c) halite (100) surfaces.

Sakuma et al., *Molecular Simulations*, 30, 861-871 (2004)

## Modeling of Friedel's Salt $[\text{Ca}_2\text{Al}(\text{OH})_6]\text{Cl}\cdot 2\text{H}_2\text{O}$ Interfaces with Aqueous $\text{Na}_2\text{SO}_4$ solution



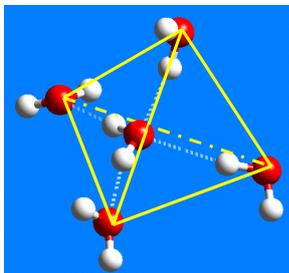
## The name is Bond, H-Bond



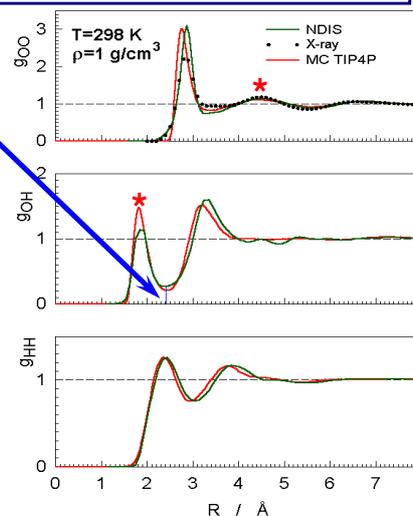
**Hydrogen bonding** is the most important phenomenon governing the properties of all aqueous systems

H --- O distance:  $< 2.45\text{\AA}$   
H—O---O angle:  $< 30^\circ$

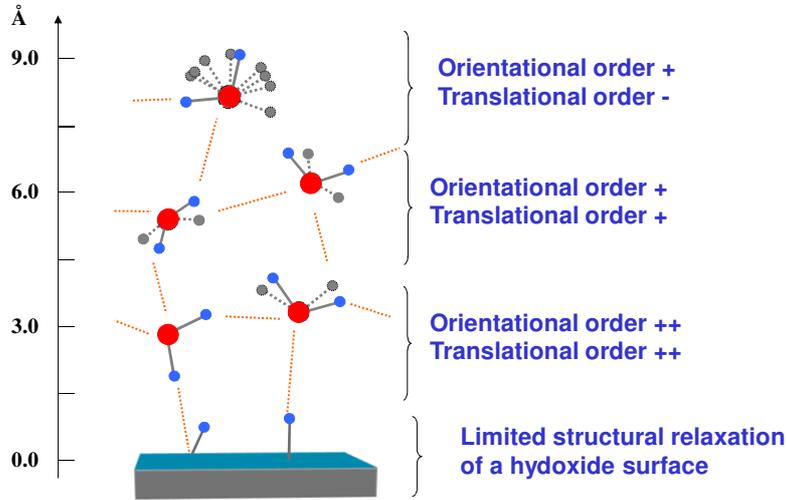
A.Luzar, *JCP*,  
113, 10663 (2000)



Ideally, each  $\text{H}_2\text{O}$  molecule in the H-bonding network can donate 2 H-bonds to its neighbors and accept 2 H-bonds from other neighbors

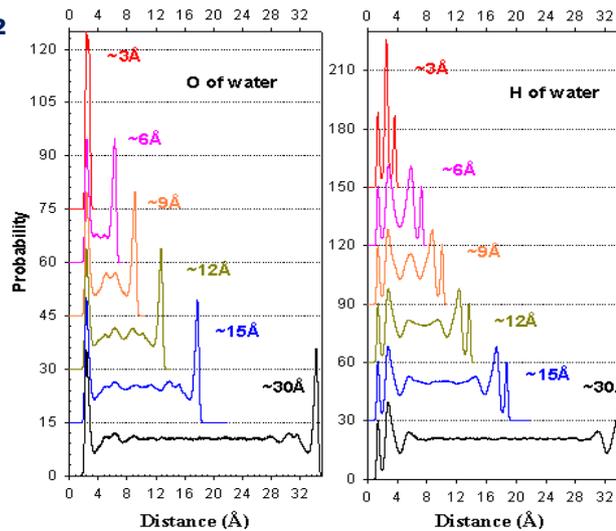


## Water at a Hydroxylated Mineral Surface, e.g., $Mg(OH)_2$ , $Ca(OH)_2$

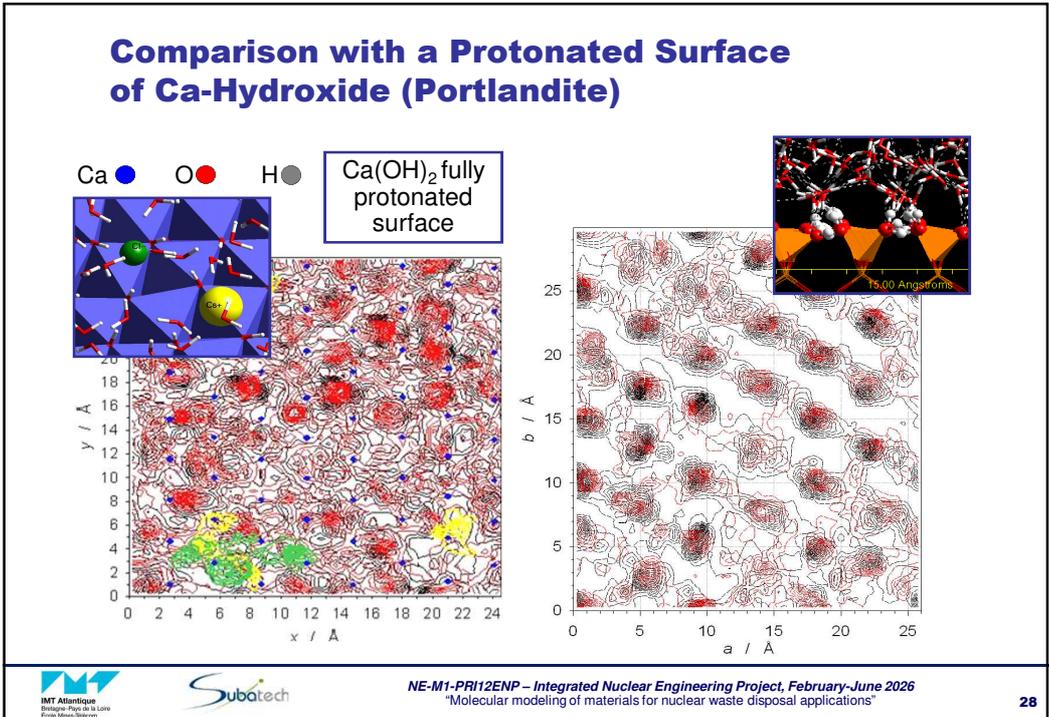
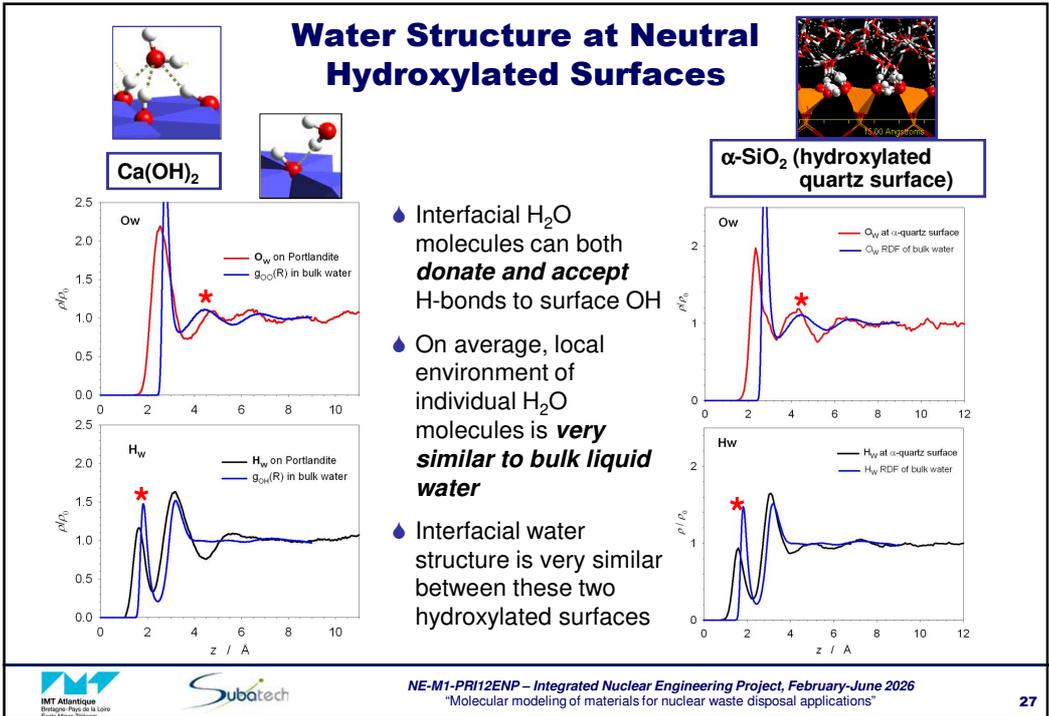


Kalinichev, Kirkpatrick, *Chemistry of Materials*, **14**, 3539-3549 (2002)  
 Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **68**, 3351-3365 (2004)

## Density Profiles of Confined Water Between Brucite $Mg(OH)_2$ Layers

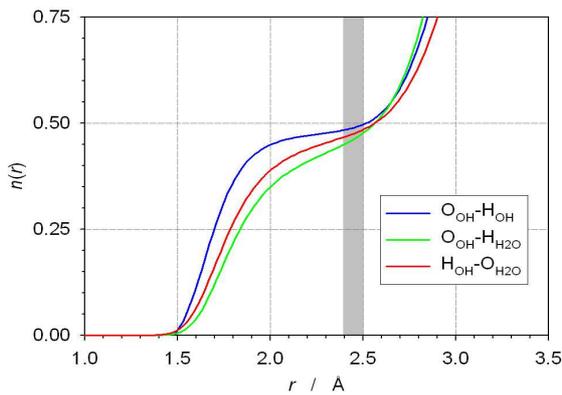


Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **68**, 3351-3365 (2004)  
 Kirkpatrick, Kalinichev, Wang, *Mineralogical Magazine*, **69**, 287-306 (2005)



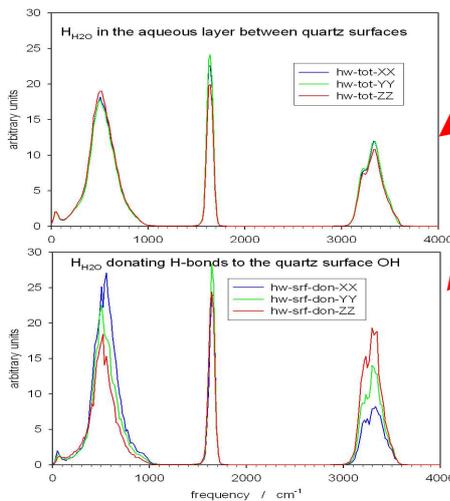
## Running Coordination Numbers and the Degree of H<sub>2</sub>O H-bonding at $\alpha$ -Quartz Surface

$$n_{ij}(r) = 4\pi\rho_i \int_0^r g_{ij}(r) r^2 dr$$



- At low pH (fully protonated quartz surface), about 50% of the surface OH groups are oriented parallel to the surface, form H-bonds to other surface OHs.
- Simultaneously, they accept H-bonds from interfacial H<sub>2</sub>O molecules.
- The other 50% of the surface OH groups are oriented nearly perpendicular to the surface and donate H-bonds to interfacial H<sub>2</sub>O.

## Velocity Autocorrelation Functions and Vibrational Spectra of Surface H<sub>2</sub>O



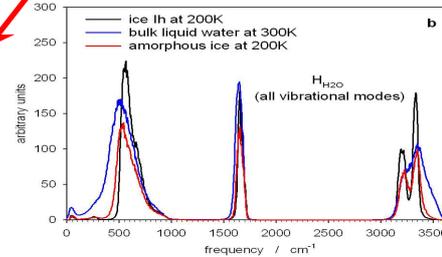
$$VACF \equiv C_{vv}(t) = \frac{\langle \mathbf{v}(0) \cdot \mathbf{v}(t) \rangle}{\langle \mathbf{v}(0)^2 \rangle} = \frac{1}{\langle \mathbf{v}(0)^2 \rangle} \int \mathbf{v}(0) \cdot \mathbf{v}(t) dT$$

$$P(\omega) = \int_0^{\infty} \frac{\langle \mathbf{v}(0) \cdot \mathbf{v}(t) \rangle}{\langle \mathbf{v}(0)^2 \rangle} \cos(\omega t) dt$$

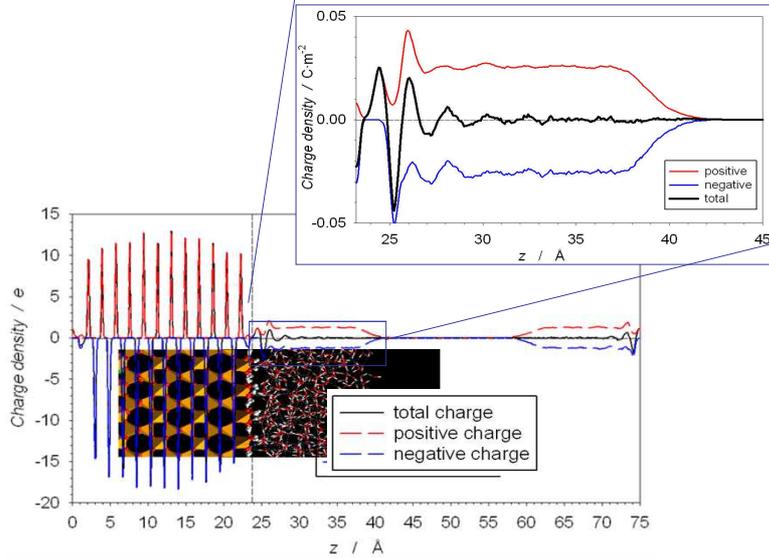
### Total H<sub>2</sub>O

### Surface H<sub>2</sub>O

- Narrower spectral bands
- More anisotropic
- Closer to "ice-like"

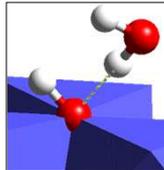
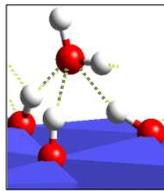
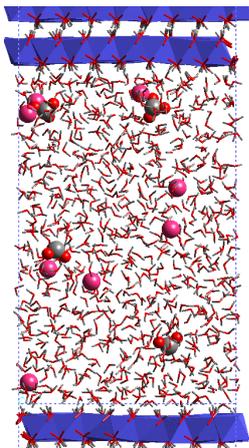


## Charge Density Profiles at Hydroxylated Quartz Surface

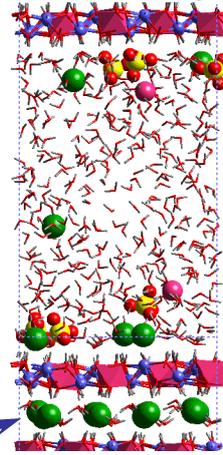
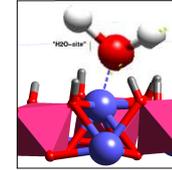
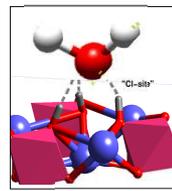


## Effect of Surface Charge on Water Structure - Metal Hydroxide Interfaces

$\text{Ca}(\text{OH})_2$   
No Structural Charge



$\text{Ca}_2\text{Al}(\text{OH})_6\text{Cl}\cdot 2\text{H}_2\text{O}$  - LDH  
Positive Structural Charge



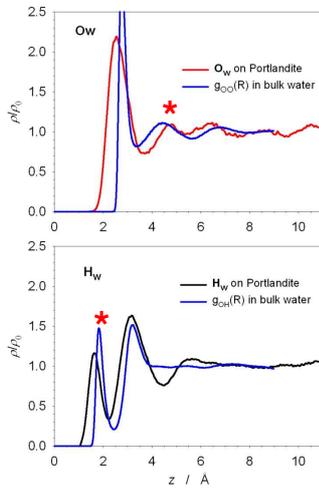
anion- $\text{H}_2\text{O}$  interlayers

Kirkpatrick, Kalinichev, Wang, *Mineral. Magazine*, **69**, 287-306 (2005)



## H<sub>2</sub>O Density Profiles and H-Bond Network at Water-Hydroxide Interfaces

### Ca-hydroxide



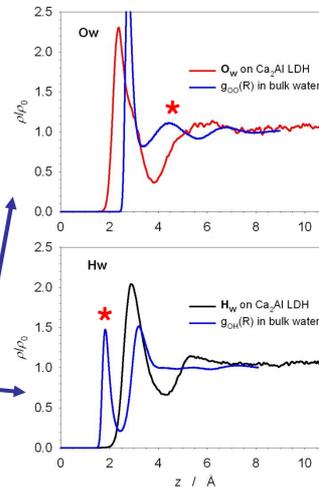
Surface H<sub>2</sub>O can both **donate and accept** H-bonds to surface OH

Local H<sub>2</sub>O environment is **similar to bulk water**

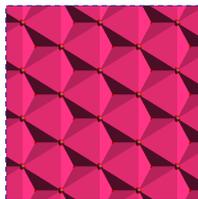
Surface H<sub>2</sub>O can **only accept** H-bonds from surface OH

Local H<sub>2</sub>O environment is **different from bulk water**

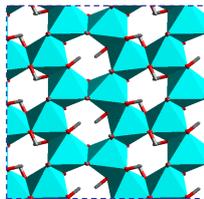
### Ca<sub>2</sub>Al LDH



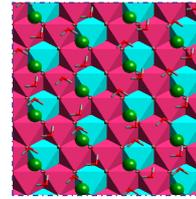
## Diverse Mineral Substrate Phases Studied



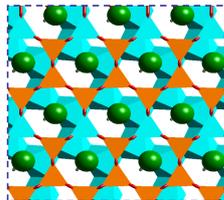
Brucite, Mg(OH)<sub>2</sub>  
Portlandite, Ca(OH)<sub>2</sub>



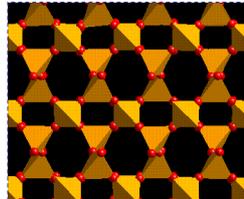
Gibbsite, Al(OH)<sub>3</sub>  
LiAl<sub>2</sub>(OH)<sub>6</sub>Cl·H<sub>2</sub>O



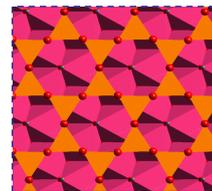
Hydrotalcite,  
Mg<sub>2</sub>Al(OH)<sub>6</sub>Cl·nH<sub>2</sub>O  
Hydrocalumite,  
Ca<sub>2</sub>Al(OH)<sub>6</sub>Cl·2H<sub>2</sub>O



Muscovite,  
KAl<sub>2</sub>(Si<sub>3</sub>,Al)O<sub>10</sub>(OH)<sub>2</sub>

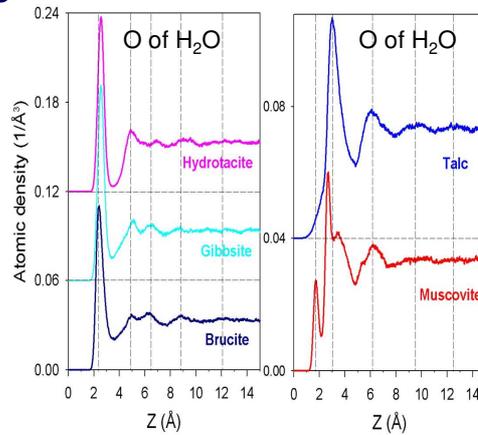
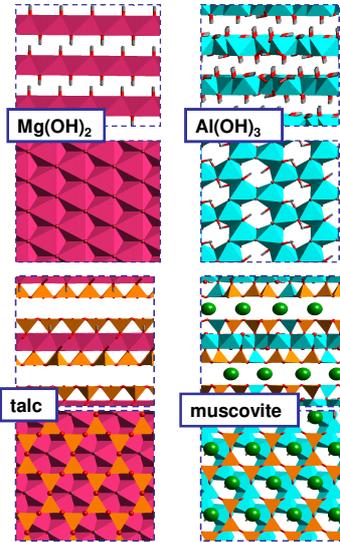


α-Quartz, SiO<sub>2</sub> (0001)  
surface fully hydroxylated



Talc, Mg<sub>3</sub>Si<sub>4</sub>O<sub>10</sub>(OH)<sub>2</sub>

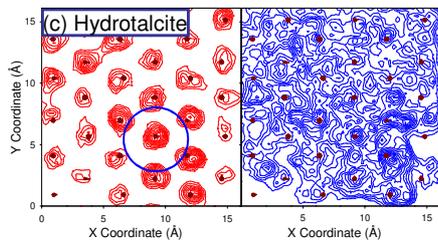
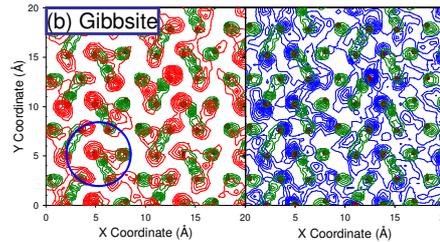
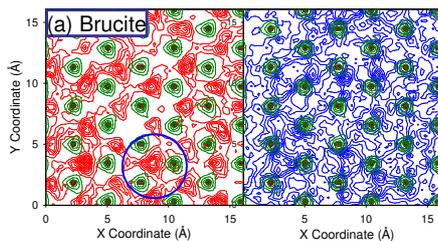
## Hydrophilic and Hydrophobic Surfaces of Oxides and Hydroxides



- ◆ The density profiles are very similar for hydroxides, but not for sheet silicates
- ◆ Unique and unambiguous interpretation is difficult with the profiles alone

Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **70**, 562-582 (2006)

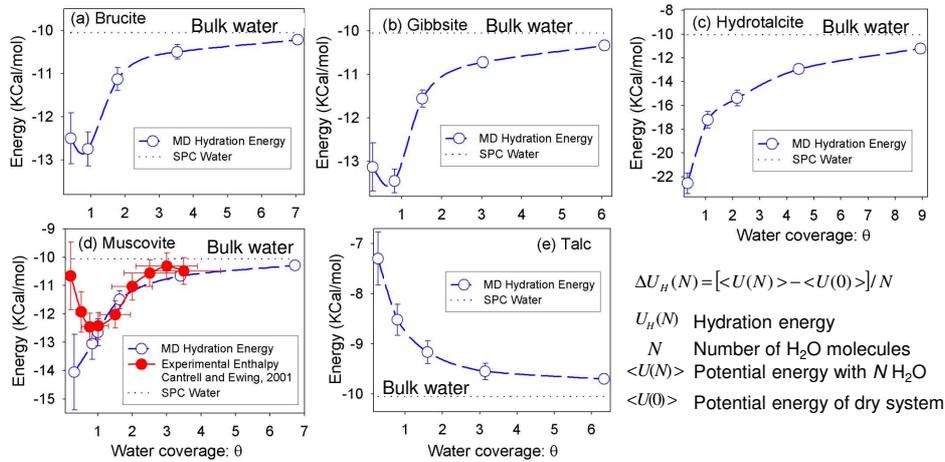
## Metal Hydroxide Surfaces



- ◆ The positions and orientations of first layer of  $H_2O$  molecules are well controlled by the substrate crystal structure, which facilitates the H-bonding with surface  $H_2O$  molecules
- ◆ Time-averaged positions of  $H_{H_2O}$  (blue) are much more disordered than  $O_{H_2O}$  (red)
- ◆ The second layer of water is much less ordered

Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **70**, 562-582 (2006)

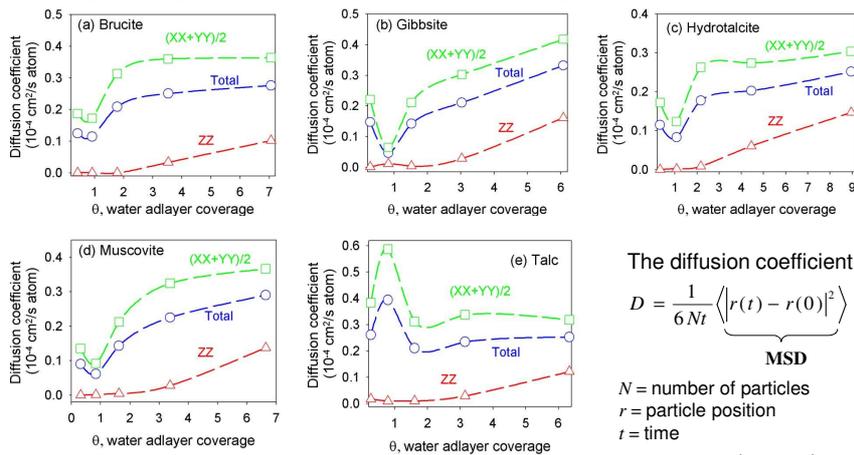
## Energetics of Surface Hydration



- For uncharged hydrophilic surfaces – the lowest hydration energy is at monolayer coverage
- For positively or negatively charged hydrophilic surfaces – the lowest hydration energy is at the lowest coverage
- For a hydrophobic surface (talc) – the lowest coverage has the highest hydration energy

Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **70**, 562-582 (2006)

## Longer Time Scale Dynamics at the Interface: Diffusion Coefficients of Interfacial Water



The diffusion coefficient:

$$D = \frac{1}{6Nt} \underbrace{\langle |r(t) - r(0)|^2 \rangle}_{\text{MSD}}$$

$N$  = number of particles  
 $r$  = particle position  
 $t$  = time

$(XX+YY)/2$ , and  $ZZ$  are the components of diffusion

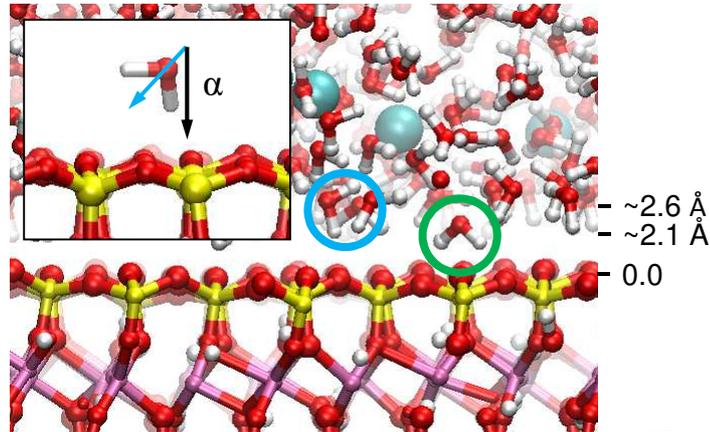
The error at each point ~10-15%

- All share similar features at hydrophilic surfaces
- A monolayer coverage stabilizes adsorbed molecules
- Hydrophobic surface: about 2 monolayer coverage stabilizes the water film

Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **70**, 562-582 (2006)

## H<sub>2</sub>O Orientation at the Clay-Water Interface

MD simulations show two principal orientations of H<sub>2</sub>O molecules close to the clay surface : **monodentate** and **bidentate**:

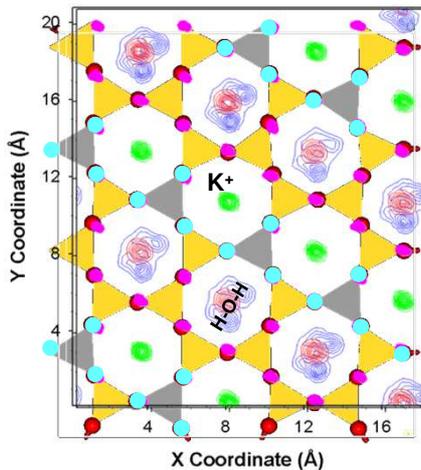


## Atomic Distributions at the Muscovite (001) Surface

Wang, Kalinichev, Kirkpatrick, Cygan, *J. Phys. Chem. B*, **109**, 15893-15905 (2005)

Wang, Kalinichev, Kirkpatrick, *Geochim. Cosmochim. Acta*, **70**, 562-582 (2006)

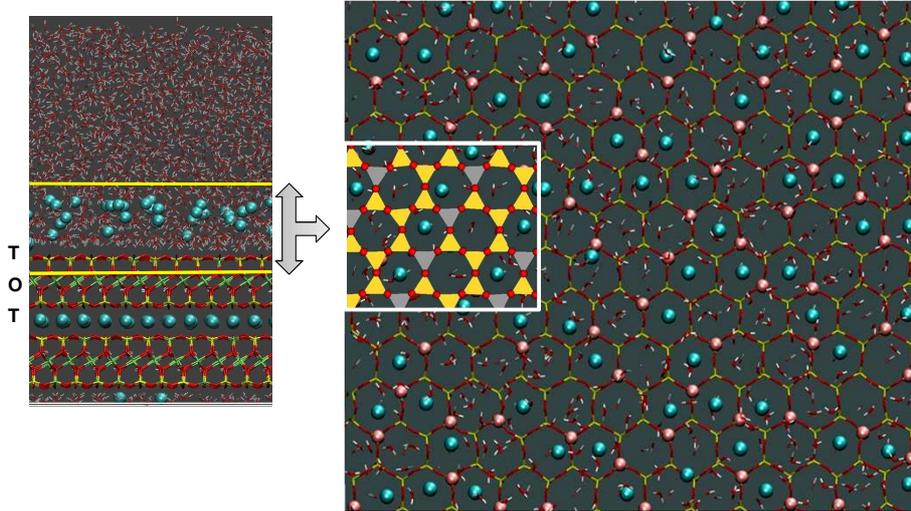
### 1-st molecular layer of surface species



- Si tetrahedron of surface layer
- Surface bridging oxygen atoms O<sub>b</sub>
- Al tetrahedral substitution
- Surface bridging oxygen atoms O<sub>bts</sub>
- Surface K<sup>+</sup> ion
- O<sub>w</sub> of adsorbed H<sub>2</sub>O
- H<sub>w</sub> of adsorbed H<sub>2</sub>O

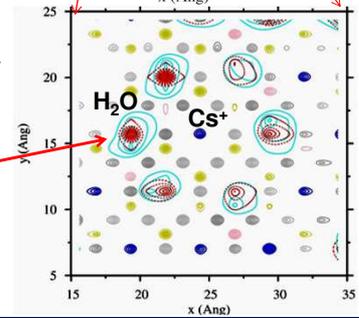
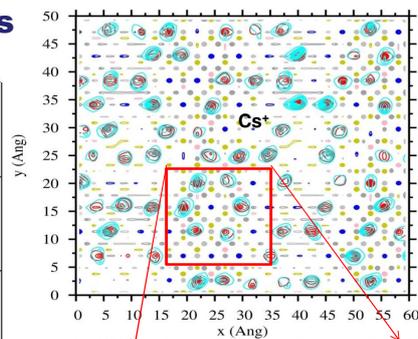
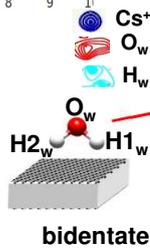
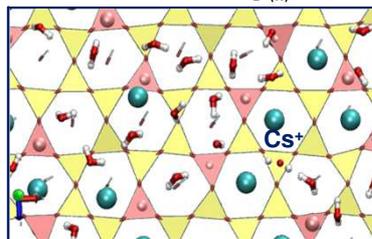
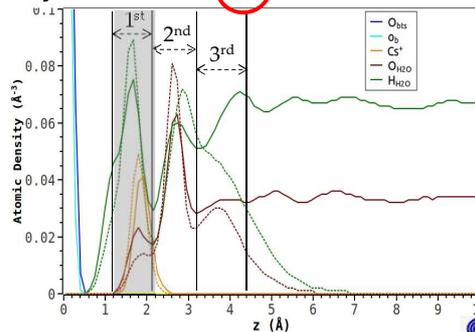
- ◆ Tetrahedral Si/Al substitutions are **ordered** in earlier models
- ◆ Hence, surface K<sup>+</sup> ions and H<sub>2</sub>O molecules are also forced to be ordered
- ◆ 50% of six-member rings are Si<sub>4</sub>Al<sub>2</sub>, the other 50% are Si<sub>5</sub>Al<sub>1</sub>.
- ◆ K<sup>+</sup> ions occupy all the Si<sub>4</sub>Al<sub>2</sub> rings

## Disorder of Al/Si Tetrahedral Sites in Muscovite



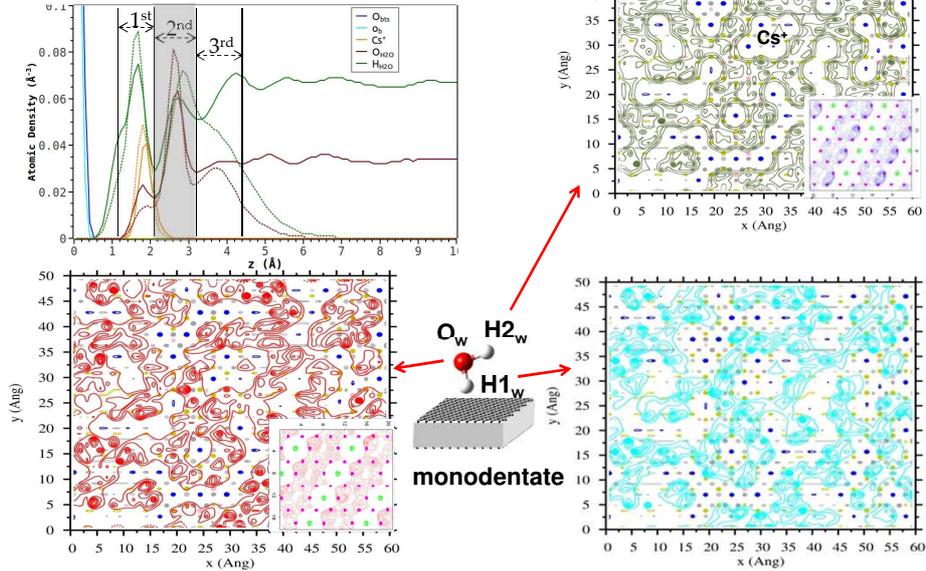
## Surface Atomic Density Maps

1-st layer of water on Cs<sup>+</sup> muscovite



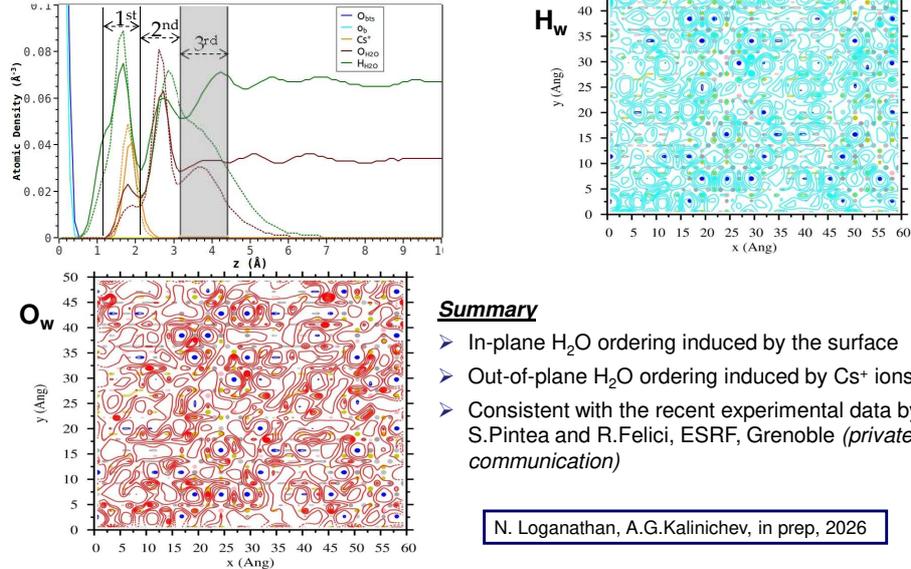
## Surface Atomic Density Maps

### 2-nd layer of water on Cs<sup>+</sup>-muscovite



## Surface Atomic Density Maps

### 3-d layer of water on Cs<sup>+</sup>-muscovite



#### Summary

- In-plane H<sub>2</sub>O ordering induced by the surface
- Out-of-plane H<sub>2</sub>O ordering induced by Cs<sup>+</sup> ions
- Consistent with the recent experimental data by S.Pintea and R.Felici, ESRF, Grenoble (*private communication*)

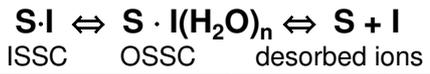
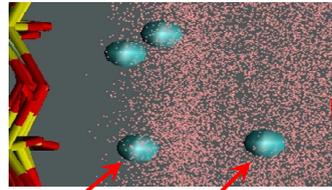
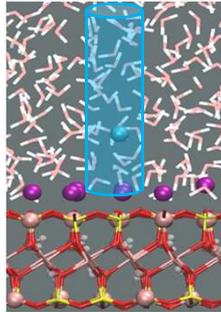
N. Loganathan, A.G.Kalinichev, in prep, 2026

## Free Energy of Ion Surface Adsorption

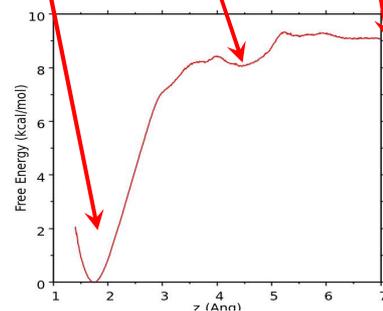
### Potential of Mean Force:

$$W(z) = -k_B T \ln(\rho(z)) + const$$

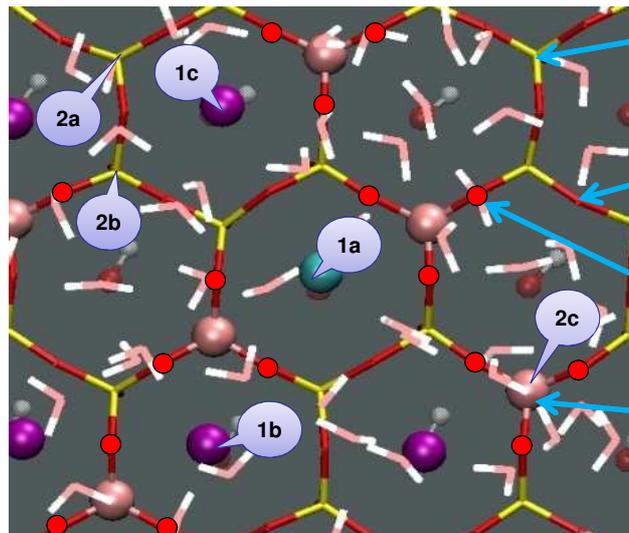
$z$  - distance from the surface



$$K_{eq} = \frac{4\pi \int \exp(-W_{ISSC}/k_B T) r^2 dr}{4\pi \int \exp(-W_{OSSC}/k_B T) r^2 dr}$$



## Different Adsorption Sites on Muscovite Surface



Si = +2.1|e|

Ob = -1.05|e|

Obts = -1.17|e|

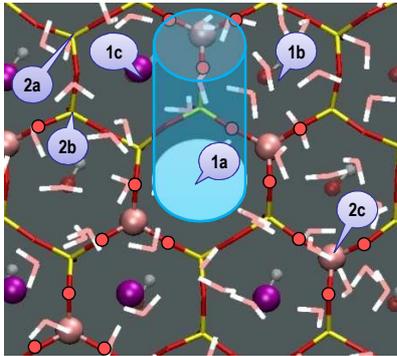
Al = +1.58|e|

## Surface Sites and Adsorption Free Energy Profiles of Cs<sup>+</sup> and K<sup>+</sup> on Muscovite

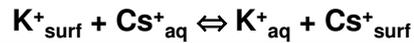
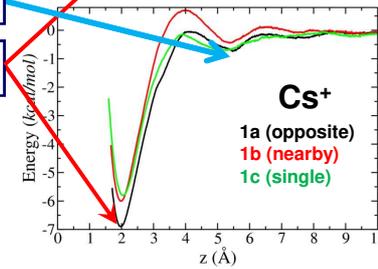
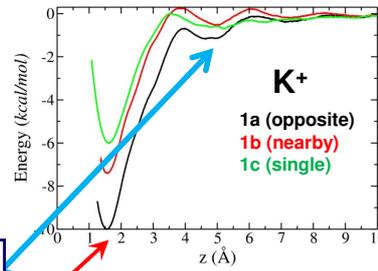
### Potential of Mean Force:

$$\text{PMF} \equiv W(z) = -k_B T \ln(\rho(z)) + \text{const}$$

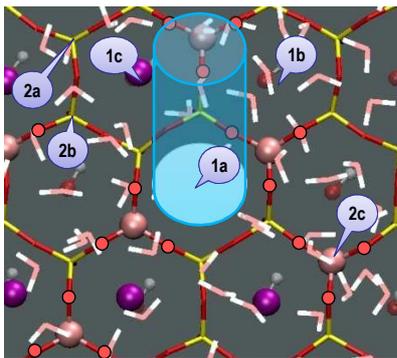
$z$  - distance from the surface



N. Loganathan, A.G.Kalinichev (2017)  
*J. Phys. Chem. C*, 121, 7829–7836

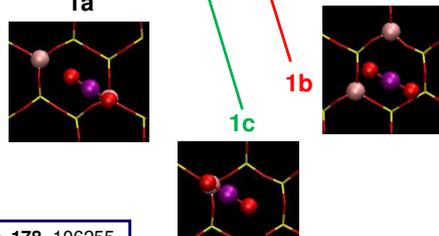
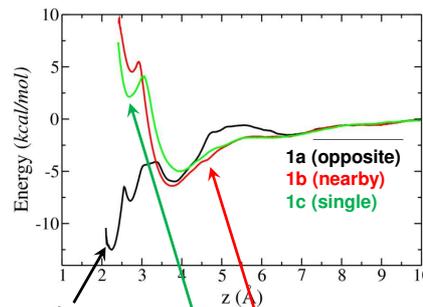


## Surface Sites and Adsorption Free Energy Profiles of UO<sub>2</sub><sup>2+</sup> on Muscovite

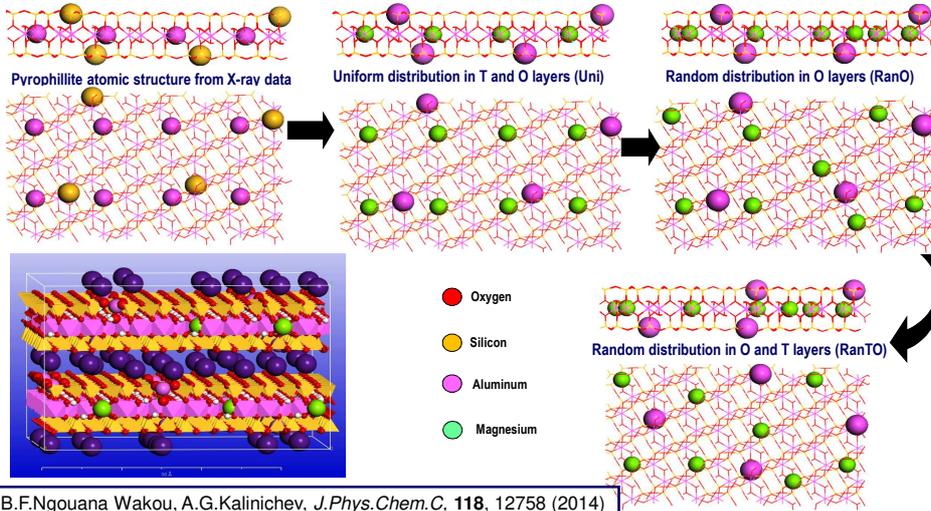


$$K_{\text{eq}} = \frac{4\pi \int \exp(-W_{\text{ISSC}}/k_B T) r^2 dr}{4\pi \int \exp(-W_{\text{OSSC}}/k_B T) r^2 dr}$$

N. Loganathan, A.G.Kalinichev (2025) *Applied Clay Science*, 178, 106255

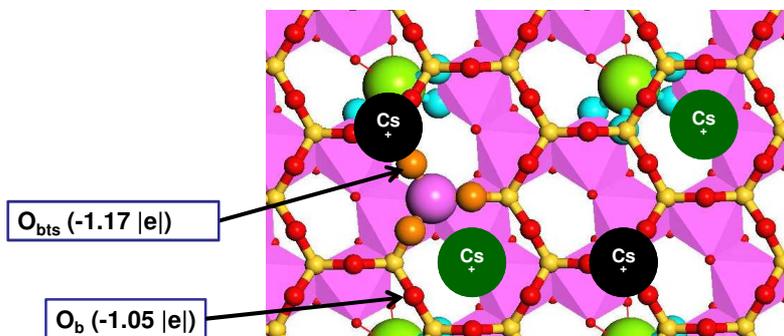
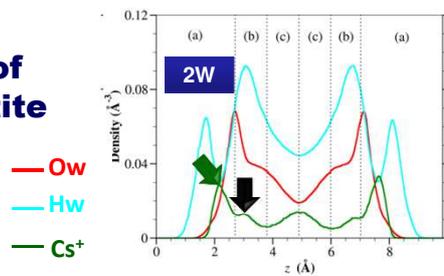


## Effects of the Charge Localization on the Adsorption and Transport of Ions in $3M^+(\text{Si}_{31}\text{Al})(\text{Al}_{14}\text{Mg}_2)\text{O}_{80}(\text{OH})_{16}\cdot n\text{H}_2\text{O}$ ( $M^+ = \text{Li}^+, \text{Na}^+, \text{K}^+, \text{Rb}^+, \text{Cs}^+, \frac{1}{2}\text{Ca}^{2+}, \frac{1}{2}\text{Sr}^{2+}, \frac{1}{2}\text{Ba}^{2+}, \frac{1}{2}\text{Ni}^{2+}$ )

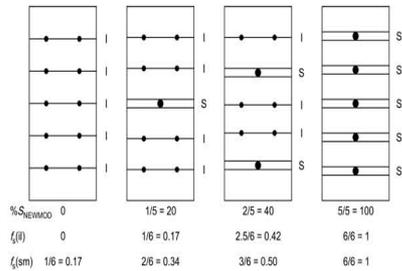


## Interlayer and Surface Adsorption Sites on (001) of $\text{Cs}^+$ -Montmorillonite Smectite

B.F. Ngouana Wakou, A.G.Kalinichev  
*J. Phys. Chem. C*, **118**, 12758-12773 (2014)



## Interstratified Illite-Smectite Structural Model and Adsorption Sites on COx Clay Particles

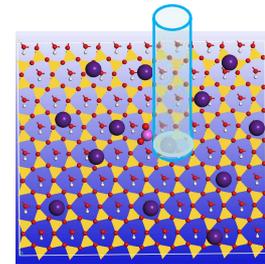
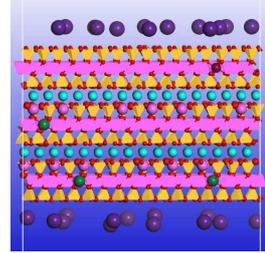


J.Srodon et al., *Clays and Clay Minerals*, **57**, 649-671 (2009)

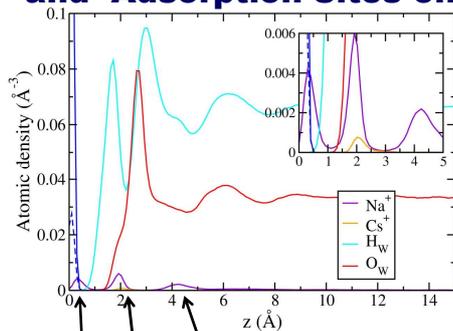
ISCz-1 model of COx used in experiments and simulations



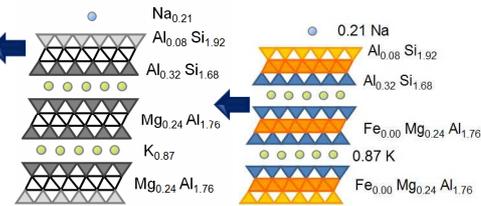
Noticeable effect not only of the distribution of charges in the nearest tetrahedral layer, but also in the lower octahedral and tetrahedral layers



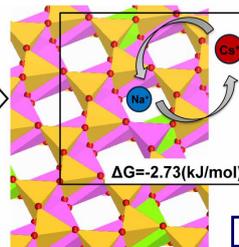
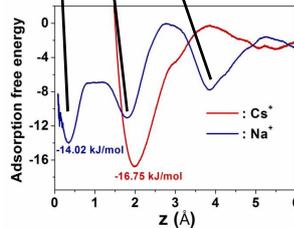
## Interstratified Illite-Smectite Structural Model and Adsorption Sites on COx Clay Particles



ISCz-1 model of COx used in experiments and simulations



J.Srodon et al., *Clays and Clay Minerals*, **57**, 649-671 (2009)

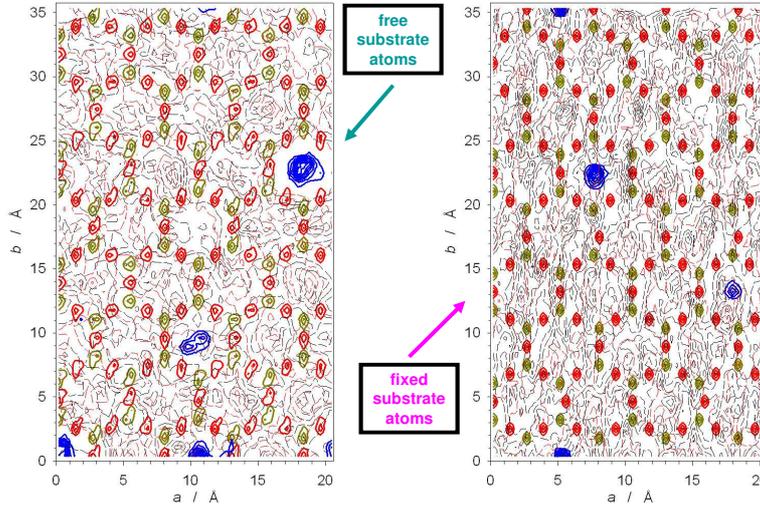


Cs<sup>+</sup> / Na<sup>+</sup> exchange on the basal surface of interstratified illite-smectite

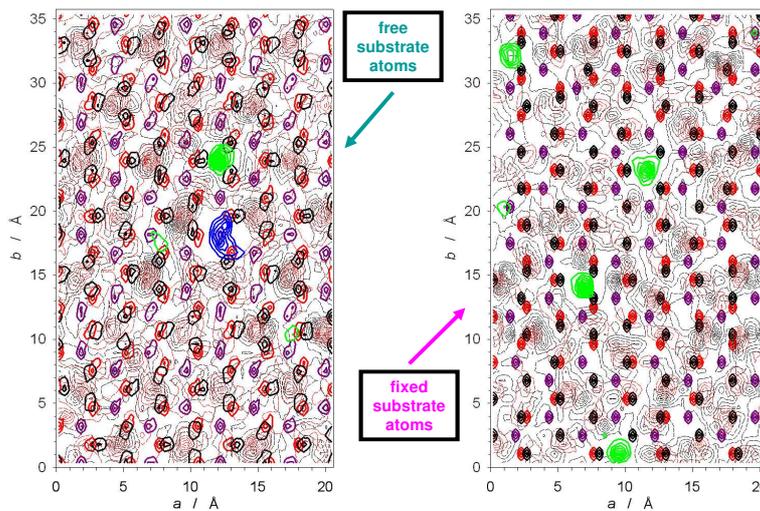
Z.Chen et al., 2026, *Env. Sci. Technol.*, in revision



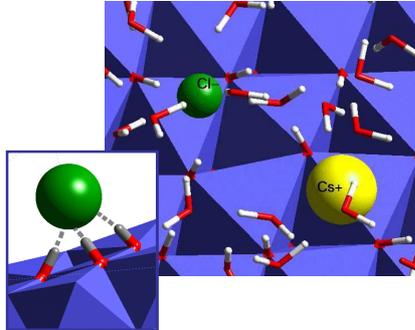
## Contour Maps of Atomic Densities within the Inner-Sphere “Siloxane” Interface of Kaolinite



## Contour Maps of Atomic Densities within the Inner-Sphere “Gibbsite” Interface of Kaolinite



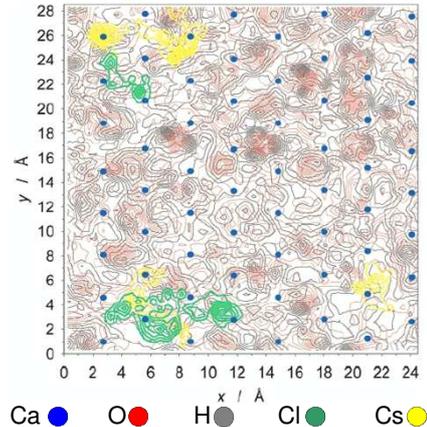
## Adsorption of Cs<sup>+</sup> and Cl<sup>-</sup> Ions at the Surface of Portlandite, Ca(OH)<sub>2</sub>



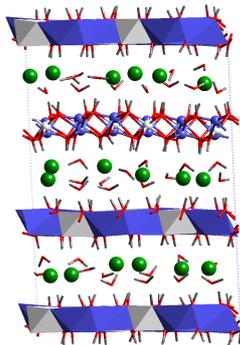
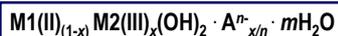
Mobility of surface OH groups is important for realistic modeling of H-bonding at the sorption sites

Kalinichev and Kirkpatrick, *Chem. Materials*, 14, 3539-3549 (2002)

Atomic density maps of the adsorbed surface species ("inner sphere" complexing)



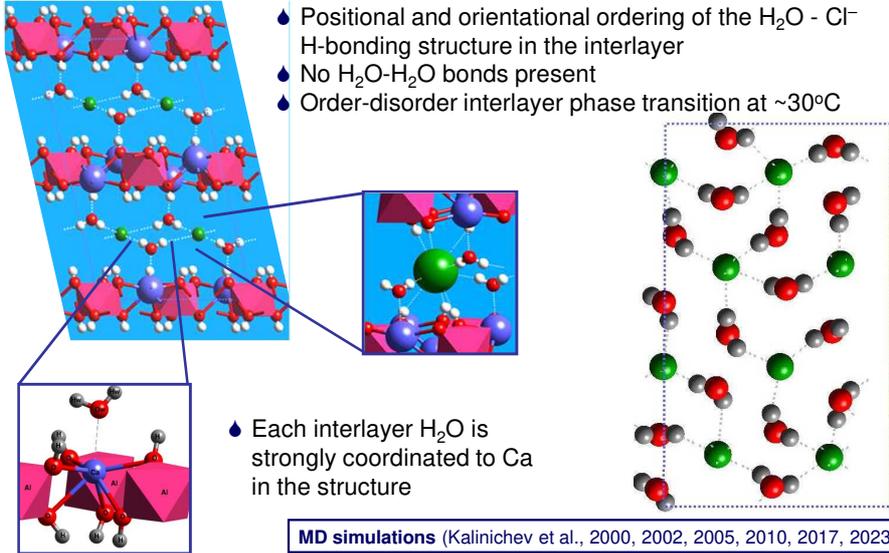
## Structure and Dynamics of Water and Ions in Layered Double Hydroxides (Anionic Clays)



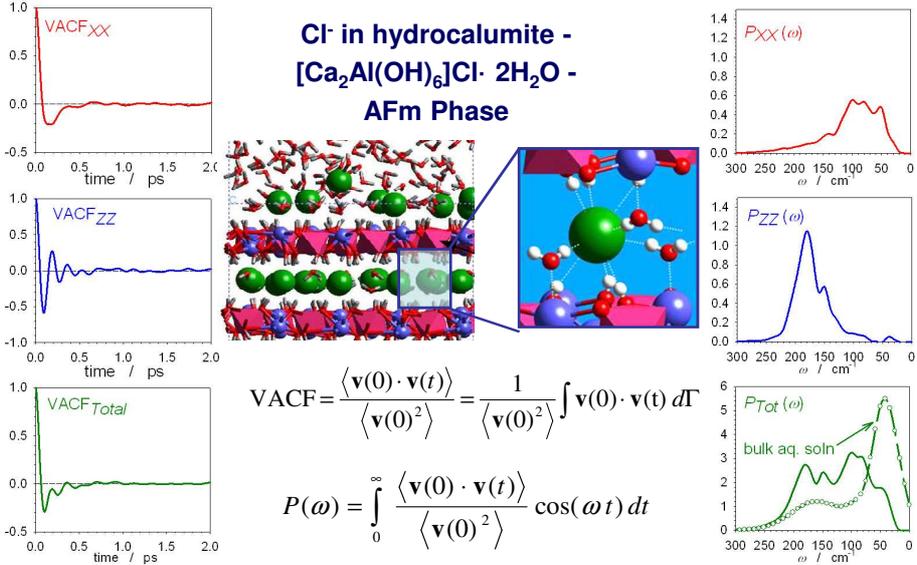
- ◆ Layered double hydroxides (LDHs), also called anionic clays
- ◆ Permanent anion exchange capacity due to isomorphous substitution of M2(III) for M1(II)
- ◆ Important applications:
  - ◆ Catalysts and catalyst supports
  - ◆ Potential repository materials
  - ◆ Drug delivery, gene therapy
  - ◆ Significant phases in cement systems
  - ◆ Origin of life
- ◆ Occur naturally, and can be easily synthesized

- ◆ Good test example of nano-confined aqueous species
- ◆ Hydrocalumite – Ca<sub>2</sub>Al(OH)<sub>6</sub>Cl·2H<sub>2</sub>O – one of the best structurally defined LDH

## Interlayer Structure of Hydrocalumite, Friedel's Salt, $[\text{Ca}_2\text{Al}(\text{OH})_6]\text{Cl} \cdot 2\text{H}_2\text{O}$



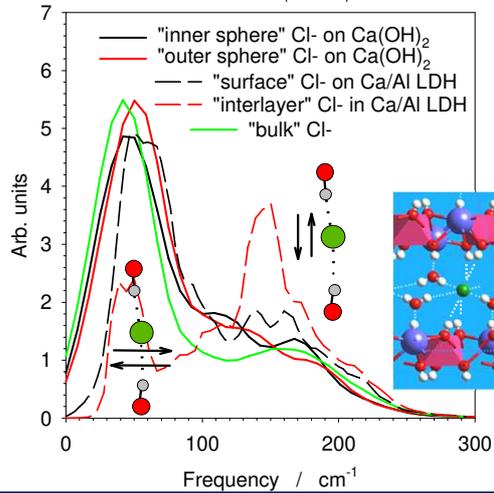
## Shorter Time Scale: Dynamics of Individual Atoms - VACFs and Vibrational Density of States



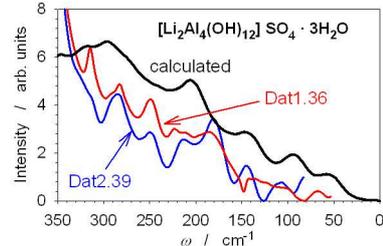
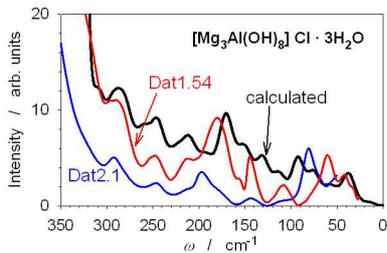
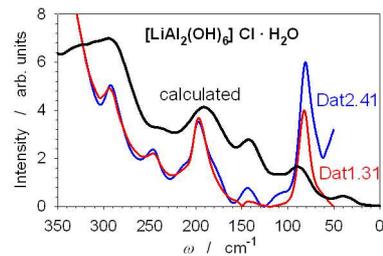
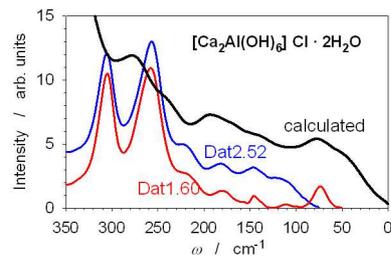
## Power Spectra of Cl<sup>-</sup> Ion Dynamics in Ca(OH)<sub>2</sub> and Ca<sub>2</sub>Al-Cl AFm phase

$$\text{VACF} \equiv C_{vv}(t) = \frac{\langle \mathbf{v}(0) \cdot \mathbf{v}(t) \rangle}{\langle \mathbf{v}(0)^2 \rangle} = \frac{1}{\langle \mathbf{v}(0)^2 \rangle} \int \mathbf{v}(0) \cdot \mathbf{v}(t) d\Gamma \quad P(\omega) = \int_0^\infty \frac{\langle \mathbf{v}(0) \cdot \mathbf{v}(t) \rangle}{\langle \mathbf{v}(0)^2 \rangle} \cos(\omega t) dt$$

- Dynamics of Cl<sup>-</sup> in the AFm interlayers is different from that on the AFm surface
- Dynamics of Cl<sup>-</sup> on the AFm surface is different from that on the Ca(OH)<sub>2</sub> surface
- Dynamics of "outer-sphere" Cl<sup>-</sup> on the Ca(OH)<sub>2</sub> surface is similar to that in the bulk aqueous solution



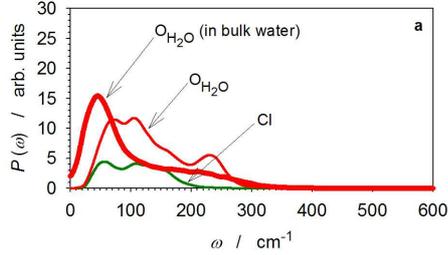
## Comparison of Observed and Computed FIR Spectra for LDH Phases



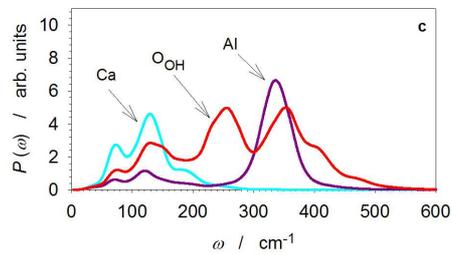
Kirkpatrick, Kalinichev, Wang et al. (2005) *Clay Minerals Society Workshop Lecture Series*, 13, 239-285.

## Low-Frequency Vibrations of Hydrocalumite

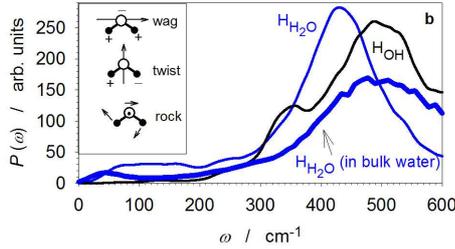
### Interlayer species – Cl<sup>-</sup>, H<sub>2</sub>O



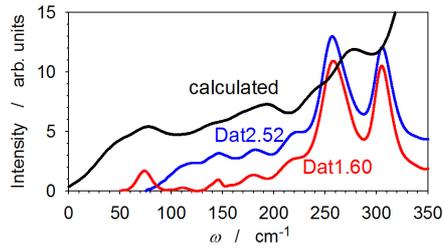
### Lattice vibrations



### H<sub>2</sub>O & OH librational modes

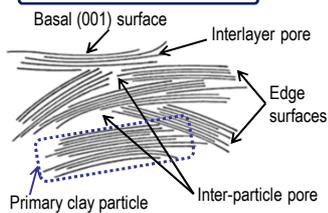


### Comparison with FIR measurements



## Particle Edges: Quantum (Ab Initio) Molecular Dynamics to Address the Chemical Reactivity

### Aggregate of clay particles



- ▶ Basal (001) surfaces and interlayers are extensively studied; their properties are reasonably well known
- ▶ Clay edges have received much less attention yet
- ▶ Ab initio (quantum) MD is a direct answer, but it is very expensive computationally
- ▶ AIMD  $\Rightarrow \sim n \times 100$  atoms;  $\sim 15 \times 15 \times 15 \text{ \AA}^3$ ;  $t \sim 10\text{-}50$  ps

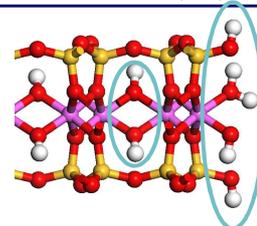
S.V.Churakov, *Geochim. Cosmochim. Acta*, **71**, 1130-1144 (2007)  
X. Liu et al., *Geochim. Cosmochim. Acta* (2012, 2013, 2014, 2015)  
S. Tazi et al., *Geochim. Cosmochim. Acta*, **94** 1-11 (2012)

### ClayFF Parametrization for Clay Edges

New special ClayFF bending terms for Mg-O-H, Al-O-H, and Si-O-H

$$U_{\text{ClayFF-MOH}} = U_{\text{ClayFF-orig}} + U_{\text{M-O-H}} = U_{\text{ClayFF-orig}} + k(\theta - \theta_0)^2$$

$k$  and  $\theta_0$  have to minimize the differences between DFT and ClayFF-MOH results

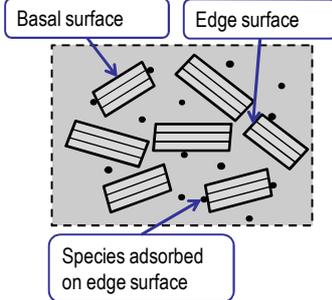


M.Pouvreau, PhD Thesis, Dec. 2016

Pouvreau, Greathouse, Cygan, Kalinichev, *J.Phys.Chem.C*, 2017, **121**, 14757-14771; 2019, **123**, 11628-11638

## ClayFF Parametrization for Clay Particle Edges

M.Pouvreau, PhD Thesis, Dec. 2016



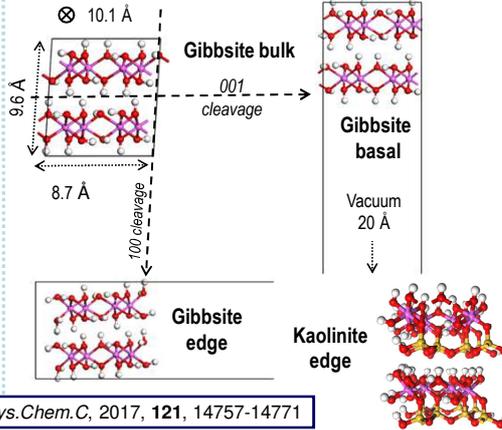
New special ClayFF M-O-H bending terms

$$U_{\text{ClayFF-MOH}} = U_{\text{ClayFF-orig}} + U_{\text{M-O-H}} = U_{\text{ClayFF-orig}} + k(\theta - \theta_0)^2$$

$k$  and  $\theta_0$  have to minimize the differences between DFT and ClayFF-MOH results

Pouvreau, Greathouse, Cygan, Kalinichev, *J.Phys.Chem.C*, 2017, **121**, 14757-14771

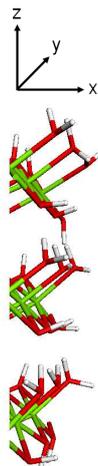
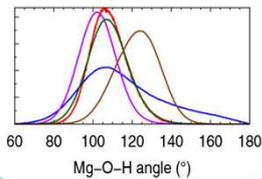
- *Brucite, gibbsite, kaolinite* edges
- 3D periodic boundary conditions
- Cells size:
  - ✓  $N_{\text{atoms}} \sim 100$  for DFT+parametrization;
  - ✓  $N_{\text{atoms}} \sim 2500$  for classical MD



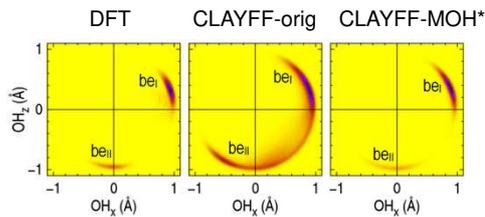
## Brucite edge: $\angle \text{Mg-O-H}$ and $\overrightarrow{\text{OH}}$

### $\angle \text{Mg-O-H}$ distribution

- DFT
- ClayFF-orig
- ClayFF-MOH:
  - $k_{\text{MgOH}} = 6 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
  - $k_{\text{AlOH}} = 15 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
  - $\theta_{\text{MgOH}} = 100^\circ$
  - $\theta_{\text{MgOH}} = 110^\circ$
  - $\theta_{\text{MgOH}} = 120^\circ$ ;  $\theta_{\text{AlOH}} = 116^\circ$



### $\overrightarrow{\text{OH}}$ distribution in xz plane



\* $\theta_0 = 110^\circ$ ;  $k_{\text{MgOH}} = 6 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$

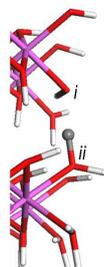
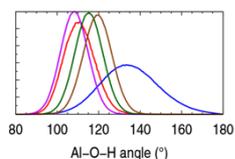
- Narrowing of  $\angle \text{Mg-O-H}$  and  $\overrightarrow{\text{OH}}$  distributions

M.Pouvreau, J.A.Greathouse, R.T.Cygan, A.G.Kalinichev, *J.Phys.Chem.C*, 2017, **121**, 14757-14771

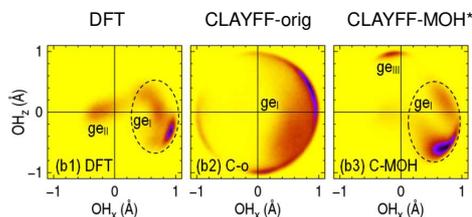
## Gibbsite Edge: $\angle \text{Al-O-H}$ and $\overline{\text{OH}}$

### $\angle \text{Al-O-H}$ distribution

- DFT
- ClayFF-orig
- ClayFF-MOH:
  - $k_{\text{MgOH}} = 6 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
  - $k_{\text{AlOH}} = 15 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
  - $\theta_{\text{MgOH}} = 100^\circ$
  - $\theta_{\text{MgOH}} = 110^\circ$
  - $\theta_{\text{MgOH}} = 120^\circ$ ;  $\theta_{\text{AlOH}} = 116^\circ$



### $\overline{\text{OH}}$ distribution of OH groups in $xz$ plane



\*  $\theta_0 = 110^\circ$ ;  $k_{\text{MgOH}} \sim 15 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$

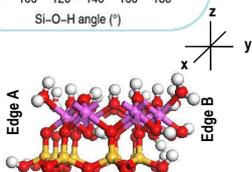
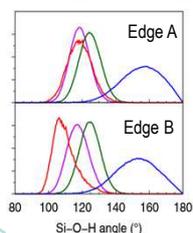
- Narrowing of  $\angle \text{Al-O-H}$  and  $\overline{\text{OH}}$  distributions
- $\angle \text{Al-O-H}$ : both  $\theta_0=100^\circ$  and  $\theta_0=110^\circ$  are close enough to DFT results
- DFT MD: observed proton transfers between OH and  $\text{OH}_2$  groups

M.Pouvreau, J.A.Greathouse, R.T.Cygan, A.G.Kalinichev, *J.Phys.Chem.C*, 2017, **121**, 14757-14771

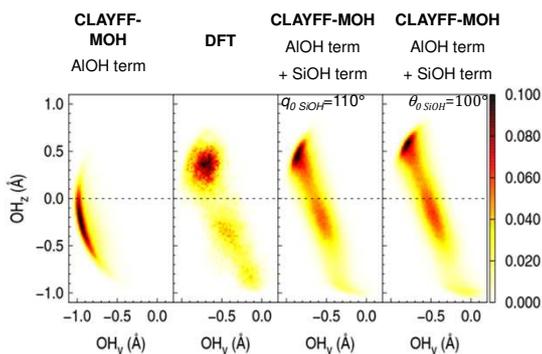
## Kaolinite Edge: $\angle \text{Si-O-H}$ and $\overline{\text{OH}}$

### $\angle \text{Si-O-H}$ distribution

- DFT
- CLAYFF-orig
- CLAYFF-MOH:
  - $k_{\text{SiOH}} = 15 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
  - $\theta_{\text{SiOH}} = 100^\circ$
  - $\theta_{\text{SiOH}} = 110^\circ$



### $\overline{\text{OH}}$ distribution of OH groups in $yz$ plane for edge A

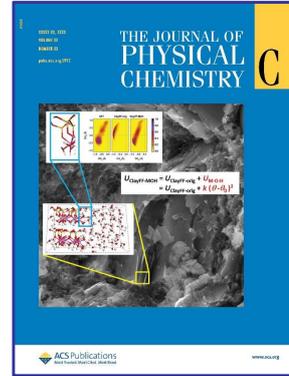
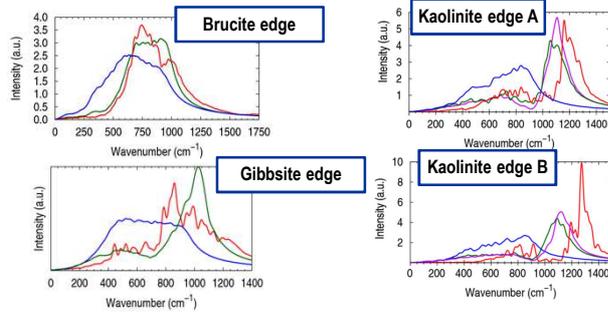


- $\theta_{\text{AIOH}} = 110^\circ$ ;  $k_{\text{AIOH}} = k_{\text{SiOH}} = 15 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
- $\theta_{\text{SiOH}} = 100^\circ$  better than  $\theta_{\text{SiOH}} = 110^\circ$  on both edges
- $\overline{\text{OH}}$  distribution not affected with both values of  $\theta_0$

M.Pouvreau et al., *J.Phys.Chem.C*, 2019, **123**, 11628-11638

## O-H Librational (M-O-H Bending) Spectra

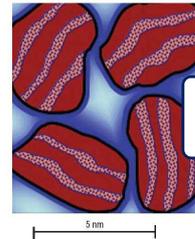
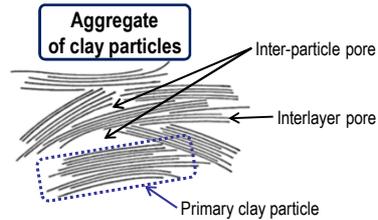
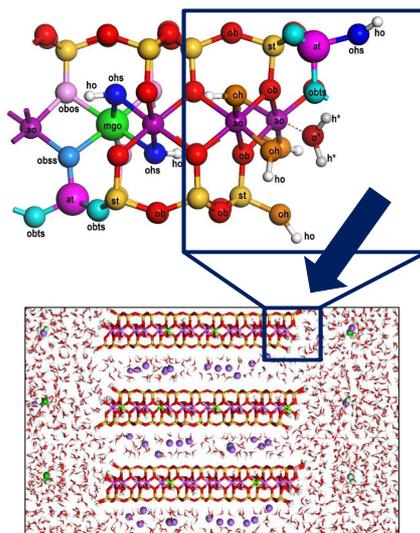
- DFT
  - CLAYFF-orig
  - CLAYFF-MOH:
    - $k_{\text{MgOH}} = 6 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
    - $k_{\text{AlOH}} = k_{\text{SiOH}} = 15 \text{ kcal}\cdot\text{mol}^{-1}\cdot\text{rad}^{-2}$
    - $\theta_{\text{MOH}} = 100^\circ$
    - $\theta_{\text{MOH}} = 110^\circ$
- Fourier Transform of the velocity autocorrelation function:
- $$P(\nu) = \int \langle \dot{\mathbf{r}}(\tau) \cdot \dot{\mathbf{r}}(t + \tau) \rangle_{\tau} e^{-i\omega t} dt$$
- We consider the velocity of the H atoms of O-H groups  
 →  $\bar{\nu} < 1500 \text{ cm}^{-1}$ : O-H libration



- Spectra shifted of 250 to 600  $\text{cm}^{-1}$  and narrowed
- Kaolinite: as expected not much difference between  $q_0=100^\circ$  and  $q_0=110^\circ$

M.Pouvreau et al.,  
*J.Phys.Chem.C*, 2019, **123**, 11628–11638

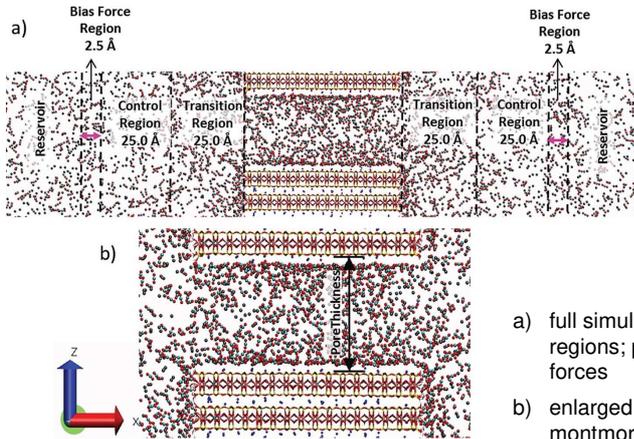
## Next: Adsorption at the Particle Edges and Mesoscale Modeling of Nanoparticle Aggregates



Aggregate of cement nanoparticles

## CH<sub>4</sub>/CO<sub>2</sub> Partitioning in Clay Nano- and Meso-Pores: Molecular Dynamics Modeling with Constant Reservoir Composition (I)

N.Loganathan, G.M.Bowers, B.F.Ngouana-Wakou, A.G.Kalinichev, R.J.Kirkpatrick, O.Yazaydin. (2019) *Physical Chemistry - Chemical Physics*, **21**, 6917-6924



Scheme of the simulation cells used in the constant reservoir composition molecular dynamics, CRC-MD calculations of CO<sub>2</sub>/CH<sub>4</sub> partitioning into pores bounded by montmorillonite basal surfaces

- a) full simulation cell showing the different regions; pink arrows represent bias forces
- b) enlarged image of the silt-like pore and montmorillonite T-O-T layers

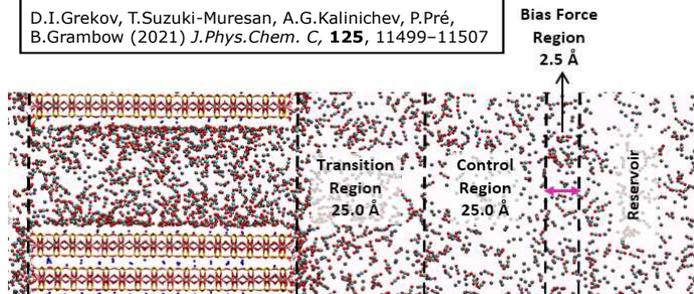
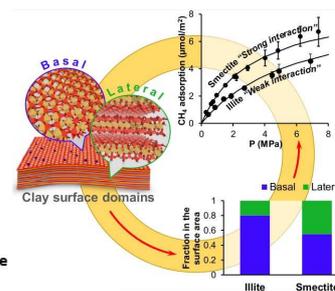
## CH<sub>4</sub>/CO<sub>2</sub> Partitioning in Clay Nano- and Meso-Pores: Molecular Dynamics Modeling with Constant Reservoir Composition (II)

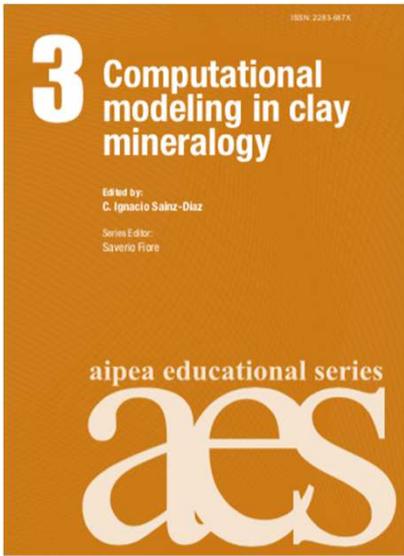
N.Loganathan, G.M.Bowers, B.F.Ngouana-Wakou, A.G.Kalinichev, R.J.Kirkpatrick, O.Yazaydin (2019) *Phys.Chem.-Chem.Phys*, **21**, 6917-6924

N.Loganathan, A.O.Yazaydin, G.M.Bowers, B.F.Ngouana-Wakou, A.G.Kalinichev, R.J.Kirkpatrick (2020) *J.Phys.Chem. C*, **124**, 2490-2500

D.I.Grekov, T.Suzuki-Muresan, A.G.Kalinichev, P.Pré, B.Grambow (2020) *Phys.Chem.-Chem.Phys*, **22**, 16727-16733

D.I.Grekov, T.Suzuki-Muresan, A.G.Kalinichev, P.Pré, B.Grambow (2021) *J.Phys.Chem. C*, **125**, 11499-11507





ISBN: 2283-9874

**3** Computational modeling in clay mineralogy

Edited by:  
C. Ignacio Sainz-Diaz

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*NE-M1-PRI12ENP – Integrated Nuclear Engineering Project, February-June 2026*  
"Molecular modeling of materials for nuclear waste disposal applications"

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**Questions for the final paper / report on one of the 21 topics, e.g., 1 – clays; 2 – cement; 3 – TiO<sub>2</sub> surfaces; 4 – corrosion; 5 – organics, etc. (Please use only as a guidance)**

- ◆ What molecular modeling method was used and why?
- ◆ Was it a fully atomistic simulations, or some simplified models were used?
- ◆ What other approximations were used in the modeling?
- ◆ What was the number of particles in the simulations? Was it big enough for the specific problem? Was it small enough to make the simulations computationally efficient?
- ◆ How long-range electrostatic interactions were handled in the simulations? Was it important for the given problem?
- ◆ What kind of boundary conditions were applied to the simulation box? Why?
- ◆ What properties of the system were calculated from the molecular simulation?
- ◆ Make a qualitative assessment of the accuracy for the calculated properties given the number of atoms in the simulated system and the duration of the simulation.
- ◆ What other properties would you additionally calculate from the same simulations for the same system?
- ◆ Formulate 2 or 3 strong points of the given molecular simulation paper and 2 or 3 weak points of the paper.




*NE-M1-PRI12ENP – Integrated Nuclear Engineering Project, February-June 2026*  
"Molecular modeling of materials for nuclear waste disposal applications"

**74**